

## Jana2006 - overview

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**Praha**



## Structural model:

- Solution and refinement of regular, disordered, twinned, modulated, composite and magnetic structures
- For description of the regular structure it allows using of anharmonic ADPs up to sixth order, multipole parameters for charge density studies and magnetic moments for magnetic structures
- For modulated and composite structures it allows modulations of all structural parameters except multipoles
- Structural parameters are automatically restricted on the base of site symmetry

### Data:

- The program can handle data single crystal and powder data from X-ray, neutron (including TOF) and electron diffraction
- Several data sets can be combined together in a joint refinement
- For X-ray data most of corrections (absorption, scaling) of integrated data can be made during data import when necessary
- Extinction correction as developed by Becker & Coppens
- Data merging and symmetry analysis is made for regular and modulated crystals

**Specify type of the file to be imported**

Single crystal:  known diffractometer formats  
 reflection file corrected for LP and absorption

**Data reduction file from:**

File name:

<input type="radio"/> <u>CAD4</u>	<input type="radio"/> <u>IPDS Stoe</u>
<input type="radio"/> <u>Nonius-CCD</u>	<input type="radio"/> <u>D9-ILL</u>
<input type="radio"/> <u>Siemens P4</u>	<input type="radio"/> <u>ILL-Vivaldi</u>
<input type="radio"/> <u>Bruker-CCD</u>	<input type="radio"/> <u>ISIS SXD</u>
<input checked="" type="radio"/> <u>Oxford Diffraction-CCD</u>	<input type="radio"/> <u>Hasylab F1</u>
<input type="radio"/> <u>Oxford Diffraction-PD</u>	<input type="radio"/> <u>Hasylab HUBER</u>
<input type="radio"/> <u>Rigaku-CCD</u>	<input type="radio"/> <u>Hasylab XDS</u>

Saclay format  Free format of th, 1, [Sig(1)]

PSI format

Debye-Scherrer method

Bragg-Brentanno method - Fixed Divergence Slit

Bragg-Brentanno method - Variable Divergence Slit

Another/unknown method

X-rays **OR** neutrons **OR** electrons

Powder data of one or more  
phases

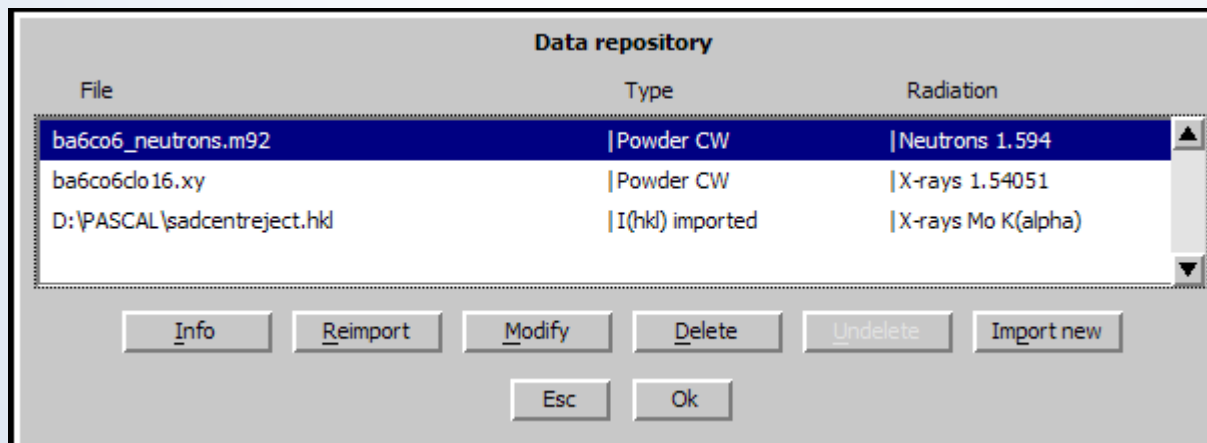
**OR**

Single crystal data of one or  
more domains

Import Wizard

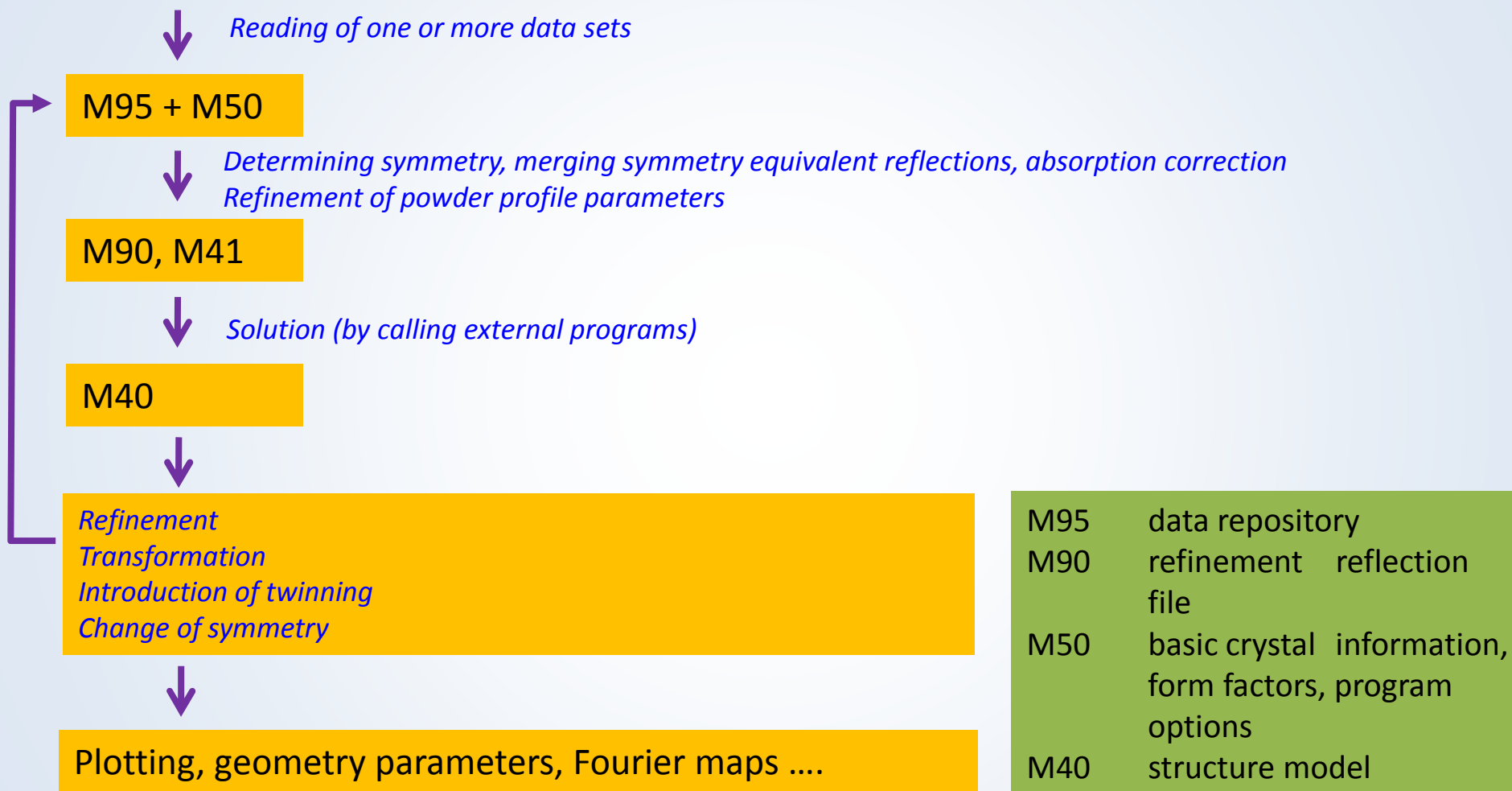
*Format conversion, cell transformation,  
sorting reflections to twin domains*

Data Repository



File	Type	Radiation
ba6co6_neutrons.m92	Powder CW	Neutrons 1.594
ba6co6do16.xy	Powder CW	X-rays 1.54051
D:\PASCAL\sadcentreject.hkl	I(hkl) imported	X-rays Mo K(alpha)

Buttons: Info, Reimport, Modify, Delete, Undelete, Import new, Esc, Ok



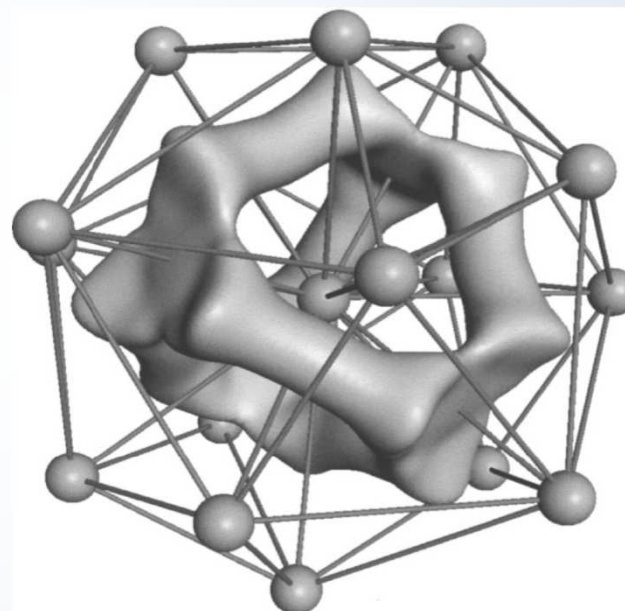
ADP tensors of the second order (harmonic ADP, “temperature parameters”) need not fully describe distribution of atoms in the crystal. Then the Gram-Charlie expansion can be used:

$$F(\mathbf{h}) = \sum_{\nu=1}^n f_{\nu}(|\mathbf{h}|) \exp(2\pi i \mathbf{r}_{\nu} \cdot \mathbf{h}) \exp(-\beta^{ij} h_i h_j) \cdot \\ \left( 1 - iC^{ijk} h_i h_j h_k + D^{ijkl} h_i h_j h_k h_l + iE^{ijklm} h_i h_j h_k h_l h_m \dots \right)$$

$C^{ijk}, D^{ijkl}, E^{ijklm}, \dots$  — contravariant components of symmetric tensor of order 3,4,5,6 are used

Ionic conductor  $\text{Ag}_8\text{TiSe}_6$  – argyrodite

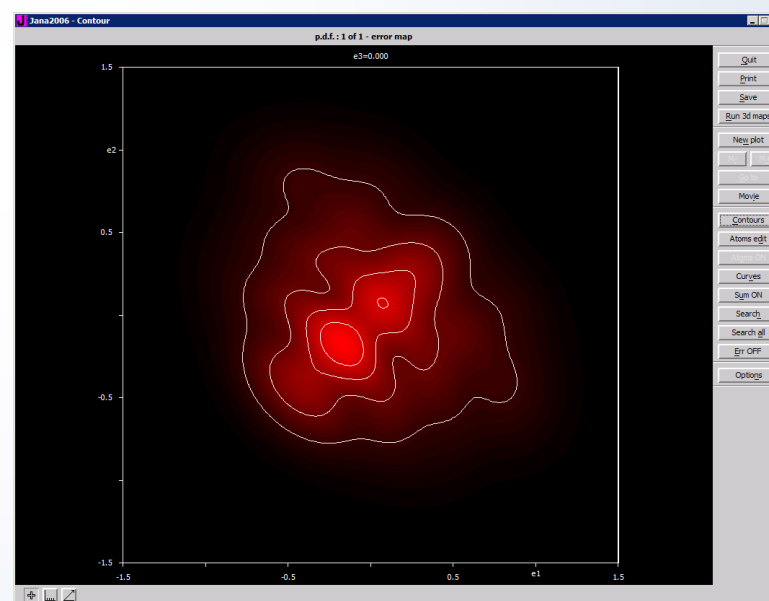
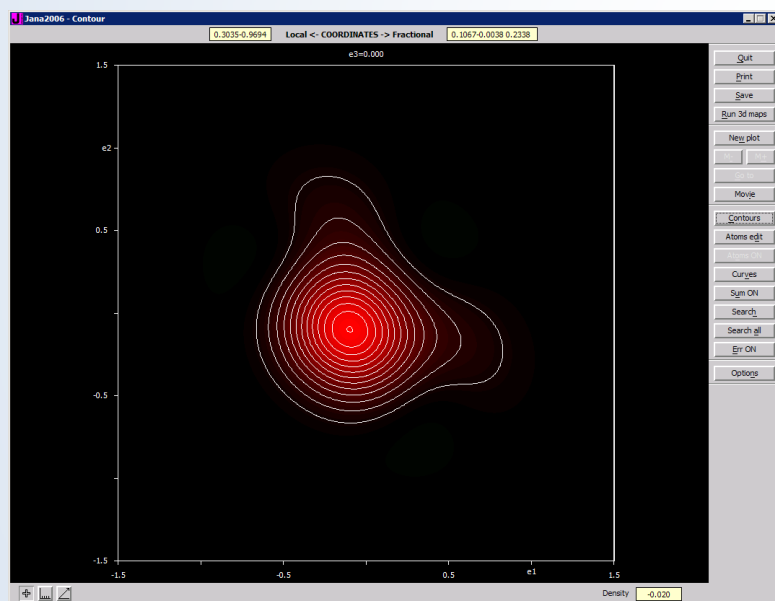
E.Gaudin, V.Petříček, F.Boucher, F.Taulelle & M.Evain, *Acta Cryst* (2000), B56, 972-979.

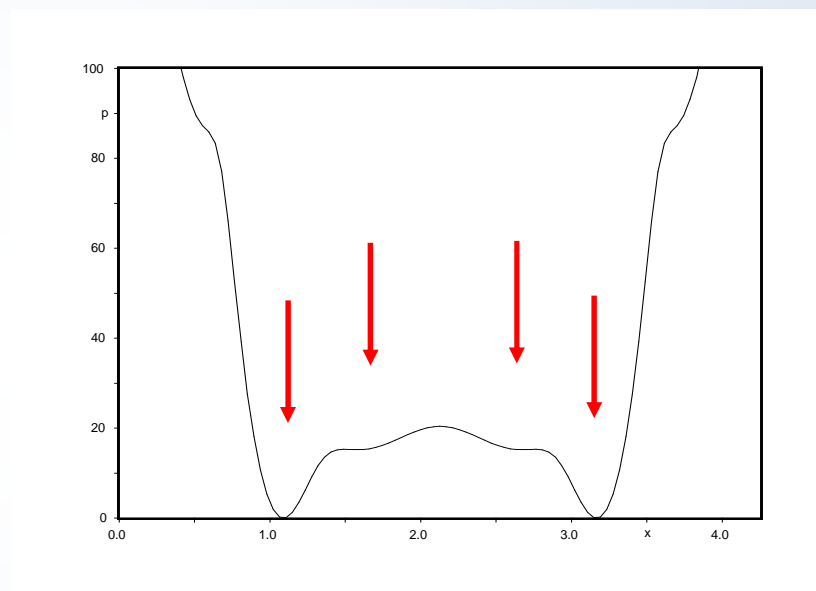
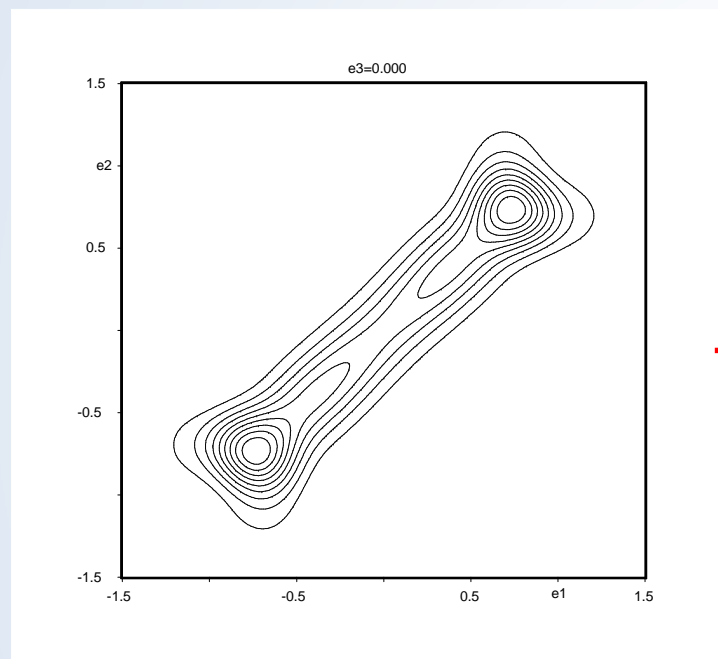


Jana calculates j.p.d.f. maps and then calls 3d visualization program.



Except of p.d.f. and j.p.d.f. map Jana can calculate also so called error map showing “map uncertainty” as follows from covariation matrix from the refinement. For its calculation the Monte-Carlo technique has been implemented (Kuks, W.F. (1992). *Acta Cryst. A***48**, 80-98):





[1,1,0] section through j.p.d.f.

Potential curve

Chance to localize possible positions of atoms after phase transition

N.K.Hansen & P.Coppens, *Acta Cryst.* (1978). **A34**, 909-921.

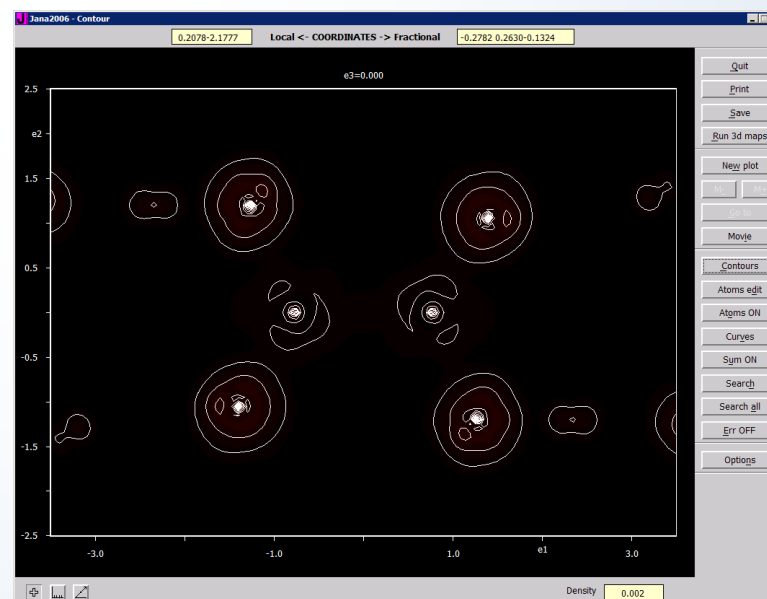
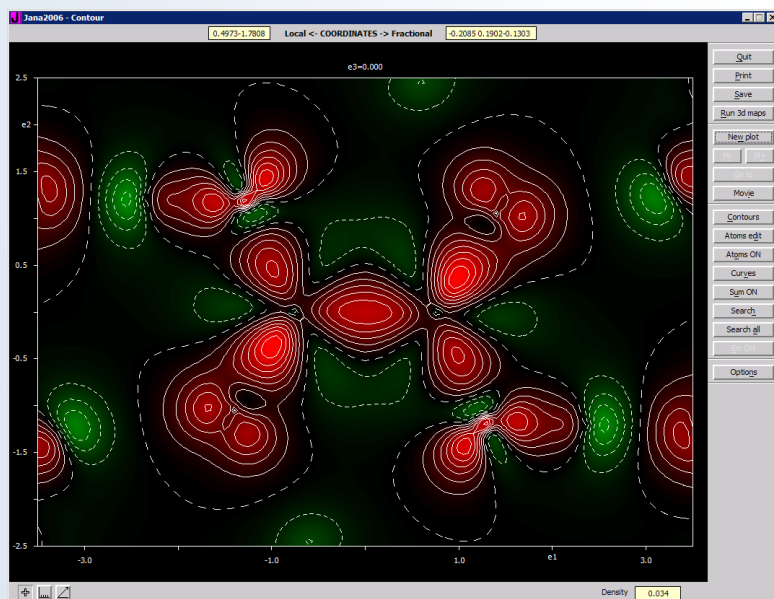
Their formalism is used in the MOLLY, XD and also in Jana2000/Jana2006

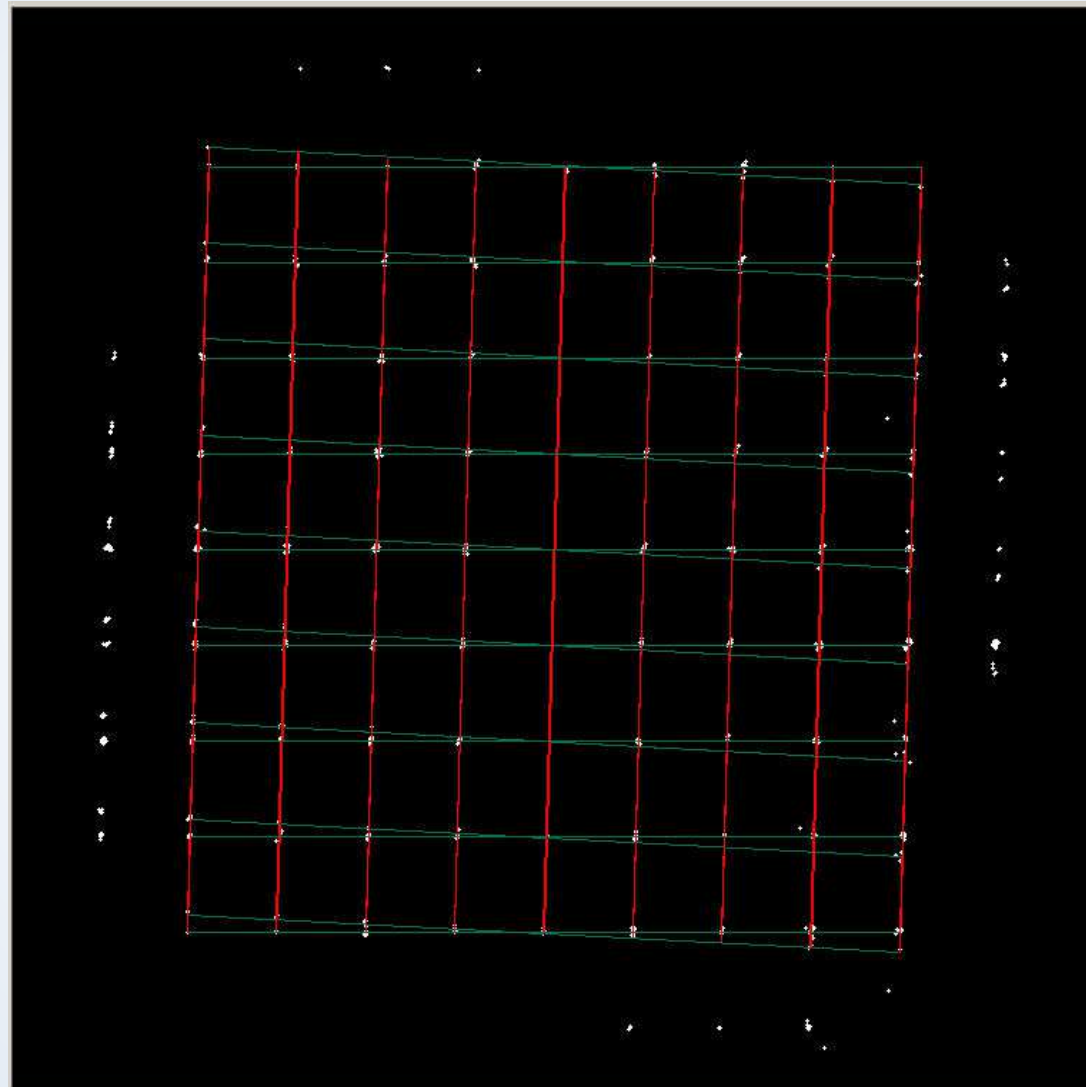
$$\rho_k(\mathbf{r}) = \rho_{k\text{core}}(r) + P_{k\text{valence}} \kappa^3 \rho_{k\text{valence}}(\kappa_k r) + \sum_l \sum_{m=-l}^l P_{klm} \kappa_k'^3 R_{kl}(\kappa_k' r) y_{lm}(\mathbf{r}/r)$$

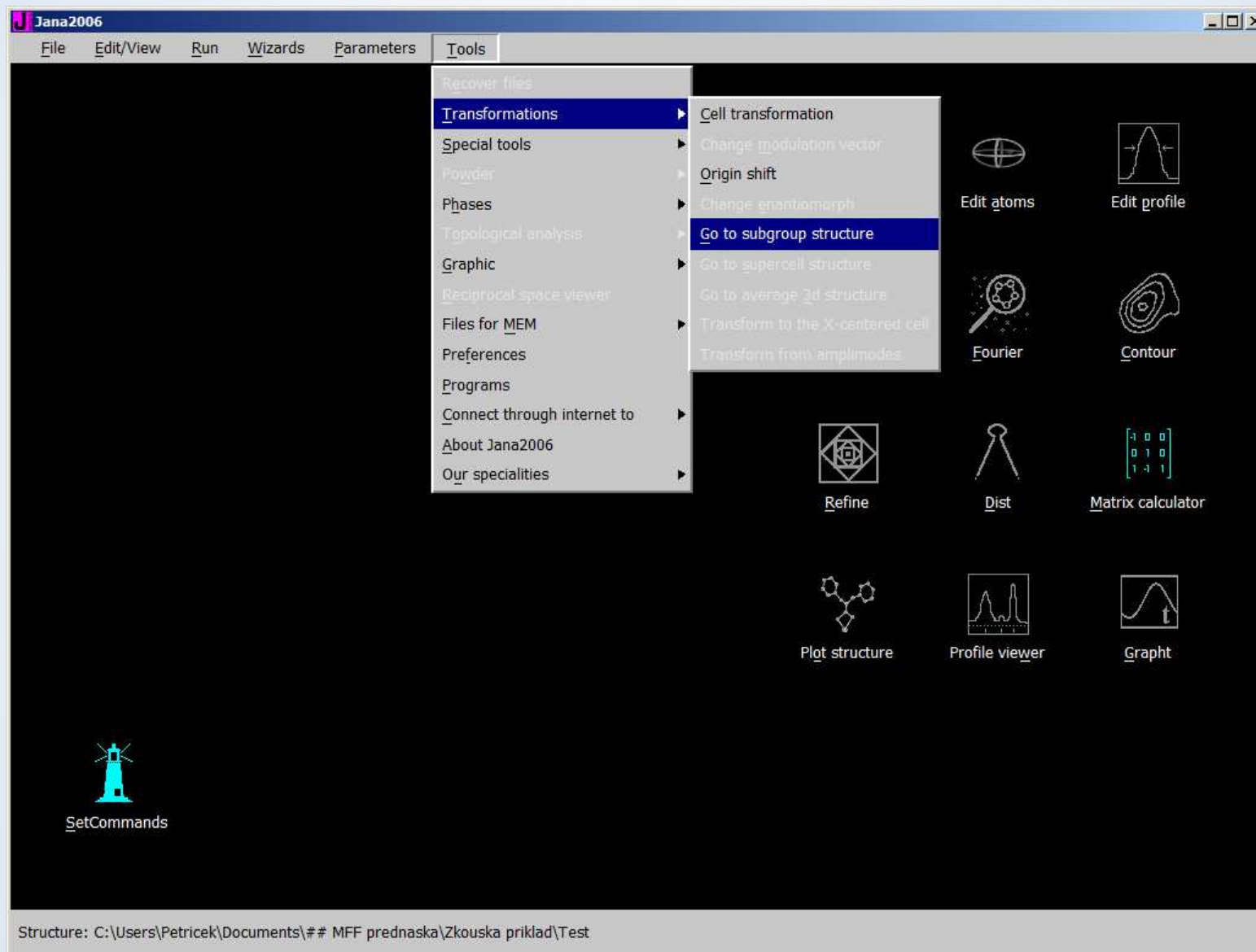
The first term is a local density connected with core electrons and the second one with valence electrons. Both have still a spherical symmetry. The second term allows spherical redistribution of valence electrons by modifying of two adjustable parameters -  $P_{k,\text{valence}}$  and  $\kappa$ .

The third term represents an aspherical term with several adjustable parameters -  $P_{klm}$  and  $\kappa_k'$ .

Similarly as for anharmonic ADP refinement the charge density maps calculated in the direct space can be compared with the error map as follows from the Monte Carlo simulation.







Jana2006

File Edit/View Run Wizards Parameters Tools

- Recover files
- Transformations**
  - Cell transformation
  - Change modulation vector
  - Origin shift
  - Go to subgroup structure**
  - Go to supercell structure
  - Go to average 3d structure
  - Transform to the X-centered cell
  - Transform from amplitudes
- Special tools
- Powder
- Phases
- Topological analysis
- Graphic
- Reciprocal space viewer
- Files for MEM
- Preferences
- Programs
- Connect through internet to
- About Jana2006
- Our specialities

Edit atoms Edit profile

Fourier Contour

Refine Dist Matrix calculator

Plot structure Profile viewer Graph

SetCommands

Structure: C:\Users\Petricek\Documents\## MFF prednaska\Zkouska priklad\Test

Jana2006

**Define subgroup**

Operator	Symbol	Direction
$x y z$	1	(0,0,0)
$-x -y -z$	-1	(0,0,0)
$x -y -z+1/2$	2	(1,0,0)
$-x y z+1/2$	c	(1,0,0)
$-x y -z+1/2$	2	(0,1,0)
$x -y z+1/2$	c	(0,1,0)
$-x -y z$	2	(0,0,1)
$x y -z$	m	(0,0,1)
$-y x z$	4	(0,0,1)
$y -x z$	4-	(0,0,1)
$y -x -z$	-4-	(0,0,1)
$-y x -z$	-4	(0,0,1)

Select non-isomorphic subgroup

---

Space group : P4/mcc      Axes : (1,0,0|0,1,0|0,0,1)      Origin : (0,0,0)

Subgroup : Cccm      Axes : (1,1,0|-1,1,0|0,0,1)      Origin : (0,0,0)      Index : 2

The selected subgroup is normal

Structure: C:\Users\Petricek\Documents\## MFF prednaska\Zkouska priklad\Test

Jana2006

**Define subgroup**

Operator	Symbol	Direction
-x y -z+1/2	2	(0,1,0)
x -y z+1/2	c	(0,1,0)
-x -y z	2	(0,0,1)
x y -z	m	(0,0,1)
-y x z	4	(0,0,1)
y -x z	4-	(0,0,1)
y -x -z	-4-	(0,0,1)
-y x -z	-4	(0,0,1)
-y -x -z+1/2	2	(1,-1,0)
y x z+1/2	c	(1,-1,0)
y x -z+1/2	2	(1,1,0)
-y -x z+1/2	c	(1,1,0)

Select non-isomorphic subgroup

---

Space group : P4/mcc      Axes : (1,0,0|0,1,0|0,0,1)      Origin : (0,0,0)

Subgroup : Cccm      Axes : (1,1,0|-1,1,0|0,0,1)      Origin : (0,0,0)      Index : 2

The selected subgroup is normal

Structure: C:\Users\Petricek\Documents\## MFF prednaska\Zkouska priklad\Test



Jana2006

**Define subgroup**

Operator	Symbol	Direction
-x y -z+1/2	2	(0,1,0)
x -y z+1/2	c	(0,1,0)
-x -y z	2	(0,0,1)
x y -z	m	(0,0,1)
-y x z	4	(0,0,1)
y -x z	4-	(0,0,1)
y -x -z	-4-	(0,0,1)
-y x -z	-4	(0,0,1)
-y -x -z+1/2	2	(1,-1,0)
y x z+1/2	c	(1,-1,0)
y x -z+1/2	2	(1,1,0)
-y -x z+1/2	c	(1,1,0)

Select non-isomorphic subgroup

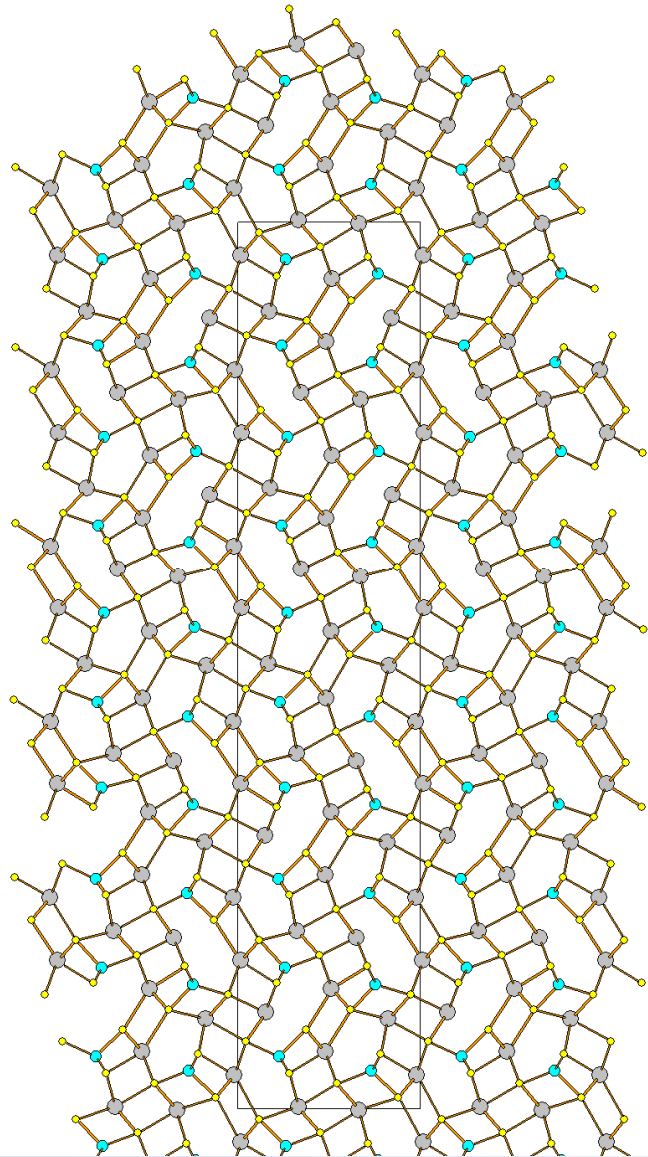
---

Space group : P4/mcc      Axes : (1,0,0|0,1,0|0,0,1)      Origin : (0,0,0)

Subgroup : Cccm      Axes : (1,1,0|-1,1,0|0,0,1)      Origin : (0,0,0)      Index : 2

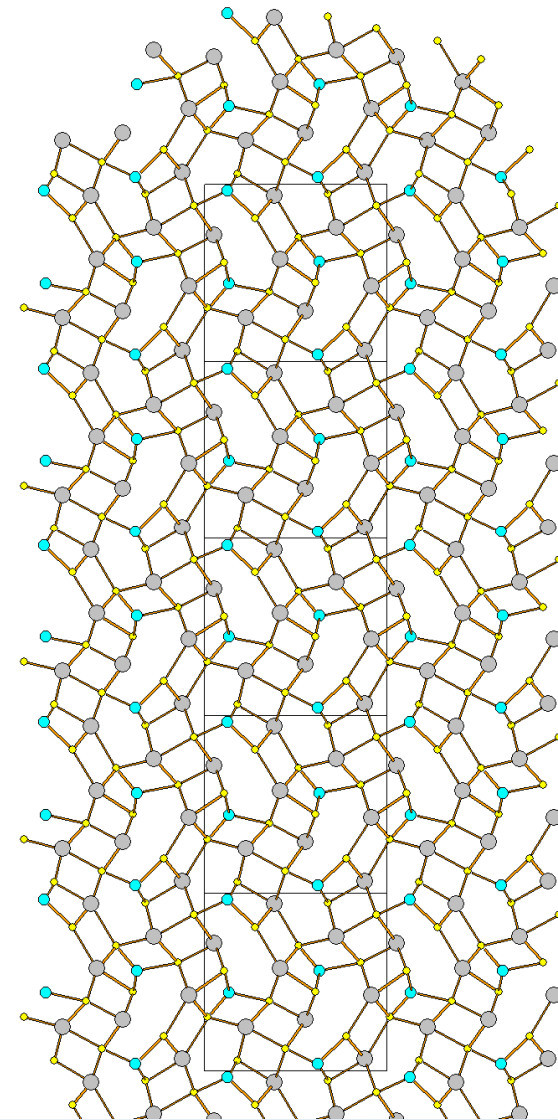
The selected subgroup is normal

Structure: C:\Users\Petricek\Documents\## MFF prednaska\Zkouska priklad\Test

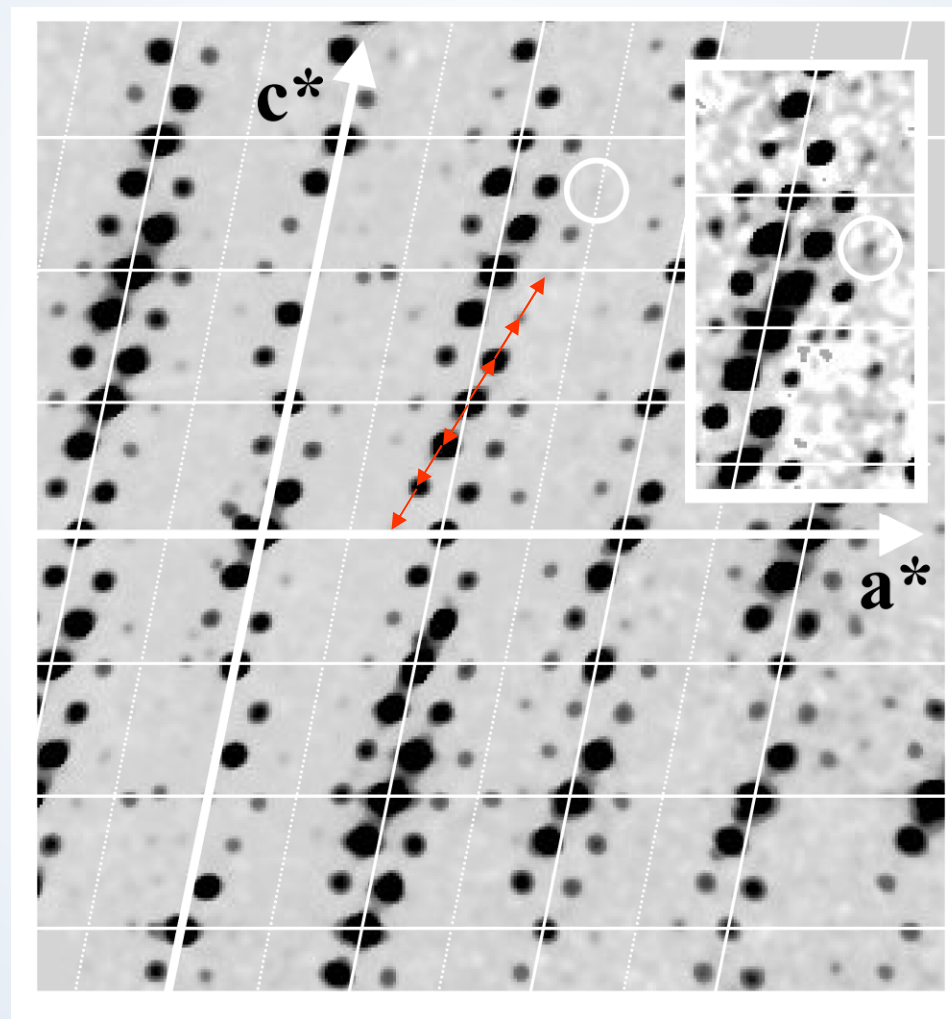


View along a  
◀ Lindströmite

View along a  
Krupkaite ▶



Diffraction pattern



Example –  $\text{Na}_2\text{CO}_3$

- One or more additional (*modulation*) vectors, have to be added to the reciprocal base in order to index all diffraction spots.
- Diffraction pattern does not have anymore 3d lattice character  $\Leftrightarrow$  the basic property of crystal 3d translation symmetry is violated but in a specific regular way.
- In the classical definition the object is not crystal  $\rightarrow$  A new definition of crystal is based on its diffraction pattern
- Modulations can considerably change physical properties of the crystal  $\rightarrow$  challenge for structure analysis to say more

## Superspace theory



Ted Janssen



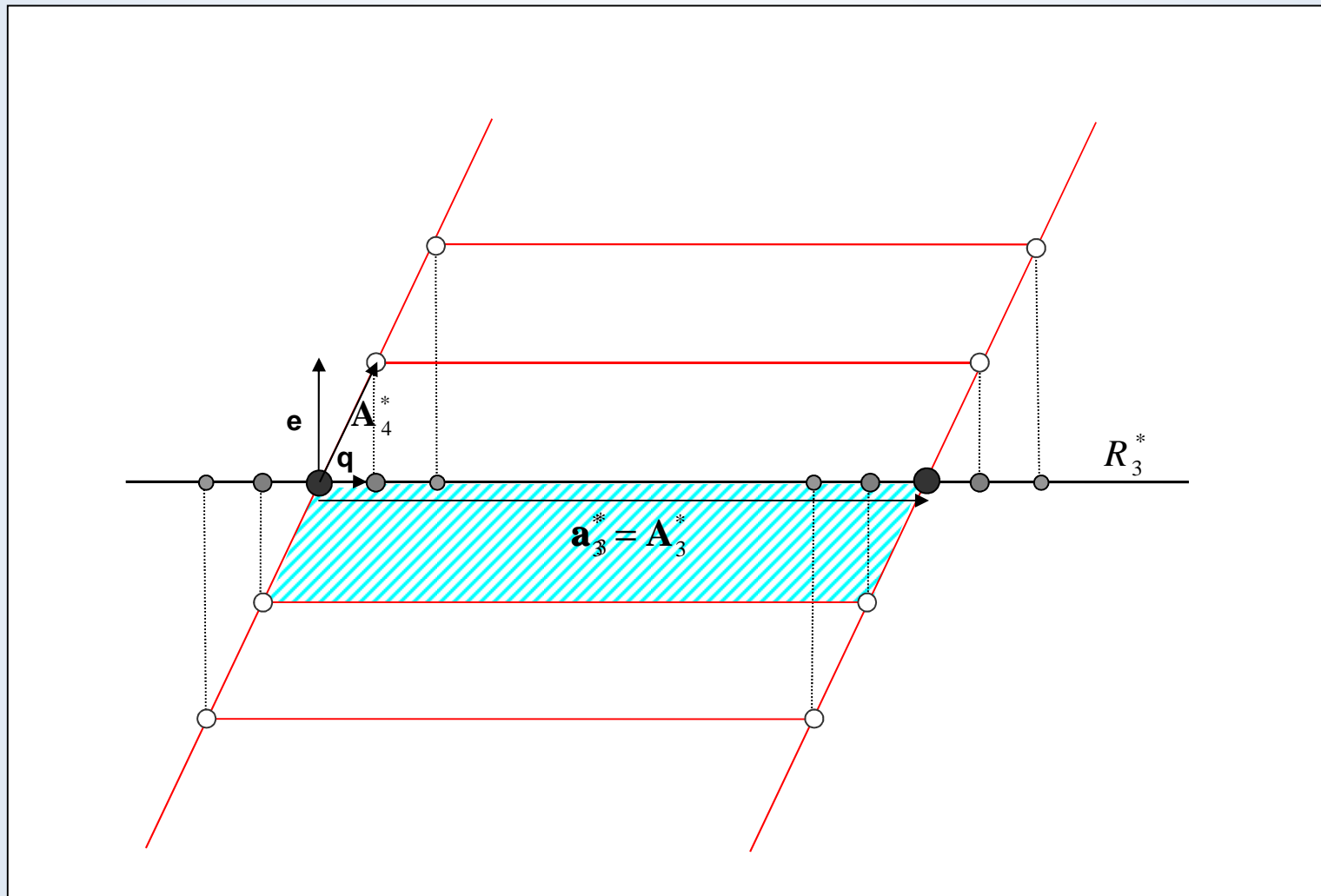
Peter M. de Wolff

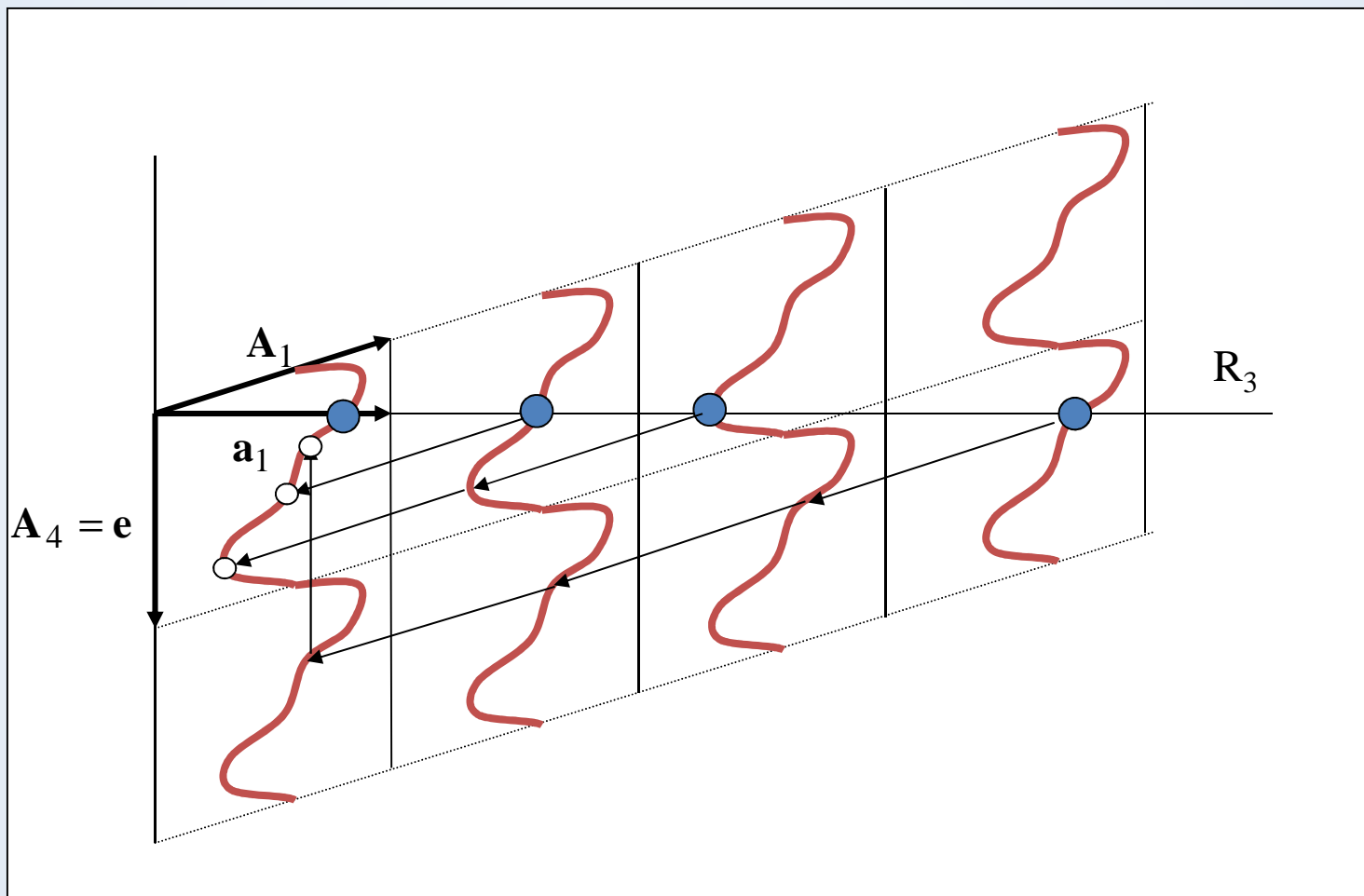


Aloysio Janner

Aminoff prize in 1998

Ewald prize in 2014





## Symmetry in the superspace

$$\tilde{\rho}(\mathbf{r}) = \tilde{\rho}\left(\mathbf{r} + \sum_{i=1}^{3+d} n_i \mathbf{A}_i\right)$$

$$\tilde{\rho}(\hat{S}\mathbf{r}) = \tilde{\rho}(\mathbf{r})$$



basic property

$$\hat{S}\mathbf{r} \cdot \hat{S}\mathbf{r} = \mathbf{r} \cdot \mathbf{r}$$



unitary operator

$$\hat{S} = (\mathbf{R}, \mathbf{t})$$



matrix representation

Trivial symmetry operator - translation symmetry :

$$\mathbf{R} = \mathbf{E}, \quad \mathbf{t} = \sum_{i=1}^{3+d} n_i \mathbf{A}_i$$



General symmetry element: 
$$\mathbf{R} = \left( \begin{array}{c|c} \Gamma_E & 0 \\ \Gamma_M & \Gamma_I \end{array} \right), \begin{bmatrix} \mathbf{s}_E \\ \mathbf{s}_I \end{bmatrix}$$

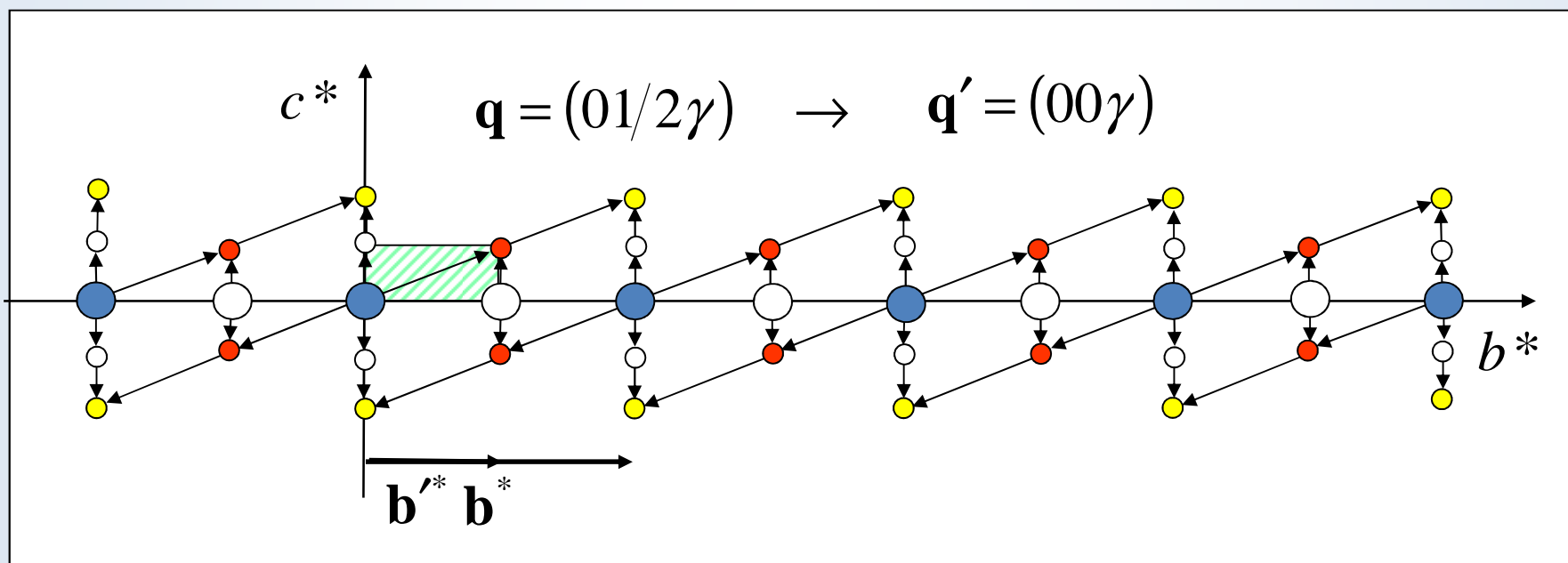
1. From the basic symmetry as determined from main reflections
2. Internal space cannot be mixed up with external one
3. Form the metric properties (unitary conditions)  $\rightarrow \mathbf{q}\Gamma_E - \Gamma_I\mathbf{q} = \Gamma_M$

The only new information is the intrinsic part of the 4<sup>th</sup> component of the translation vector which, analogically to 3d symmetry, affects systematic extinctions of reflections (here satellites). It expressed how the modulation wave is shifted the internal space.

## Super-space group symbols

Example:  $Pmna(01/2\gamma)s00$

The rational part of the modulation vector represents an additional centring. It is much more convenient to use the centred cell instead of the explicit use of the rational part of the modulation vector.



## Solution of modulated structures

Modulations can affect any of structural parameters - positions, occupancy, ADP's. The most crucial for solution is positional modulation:

$$\mathbf{r}[\mathbf{q} \cdot (\mathbf{r}_0 + \mathbf{n}) + t] = \mathbf{r}_0 + \mathbf{u}[\mathbf{q} \cdot (\mathbf{r}_0 + \mathbf{n}) + t]$$

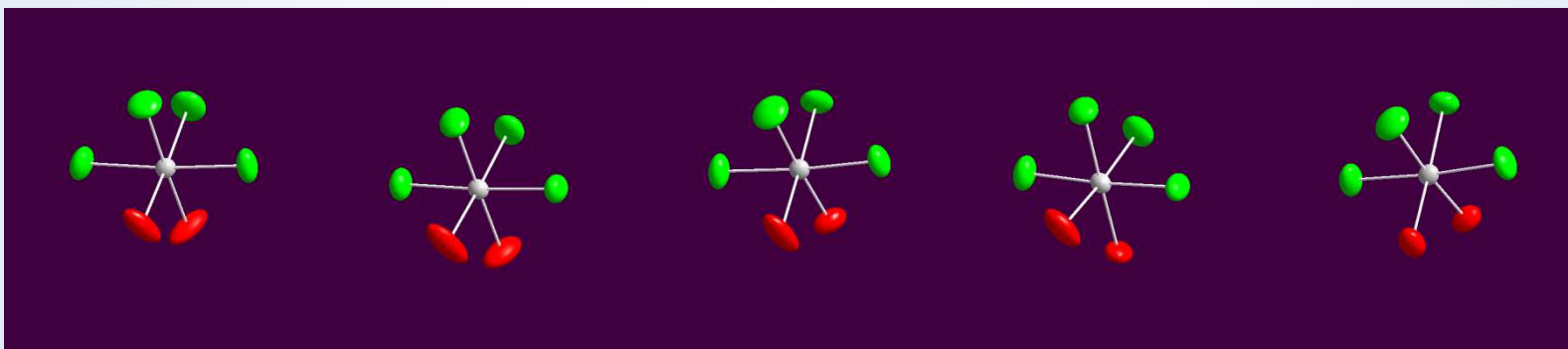
Small displacements (weak satellites)

The average structure can be solved by a standard way (direct methods, heavy atom methods) from main reflections. The modulations can be found directly from subsequent refinements. As a starting values for modulations we can use just a small starting values.

Recommended strategy - refine first modulations of atoms exhibiting some anomalies - large ADP's, split positions.

Example: J. Peterková, M. Dušek V. Petříček & J. Loub, (1998). *Acta Cryst.* B54, 809.

$\text{AsKF}_4(\text{OH})_2$  - potassium dihydroxotetrafluoroarsenate  
the anion  $\text{AsF}_4(\text{OH})_2$  strongly modulated



## Charge flipping method

A new technique applicable to regular and modulated structures,

### For regular structures:

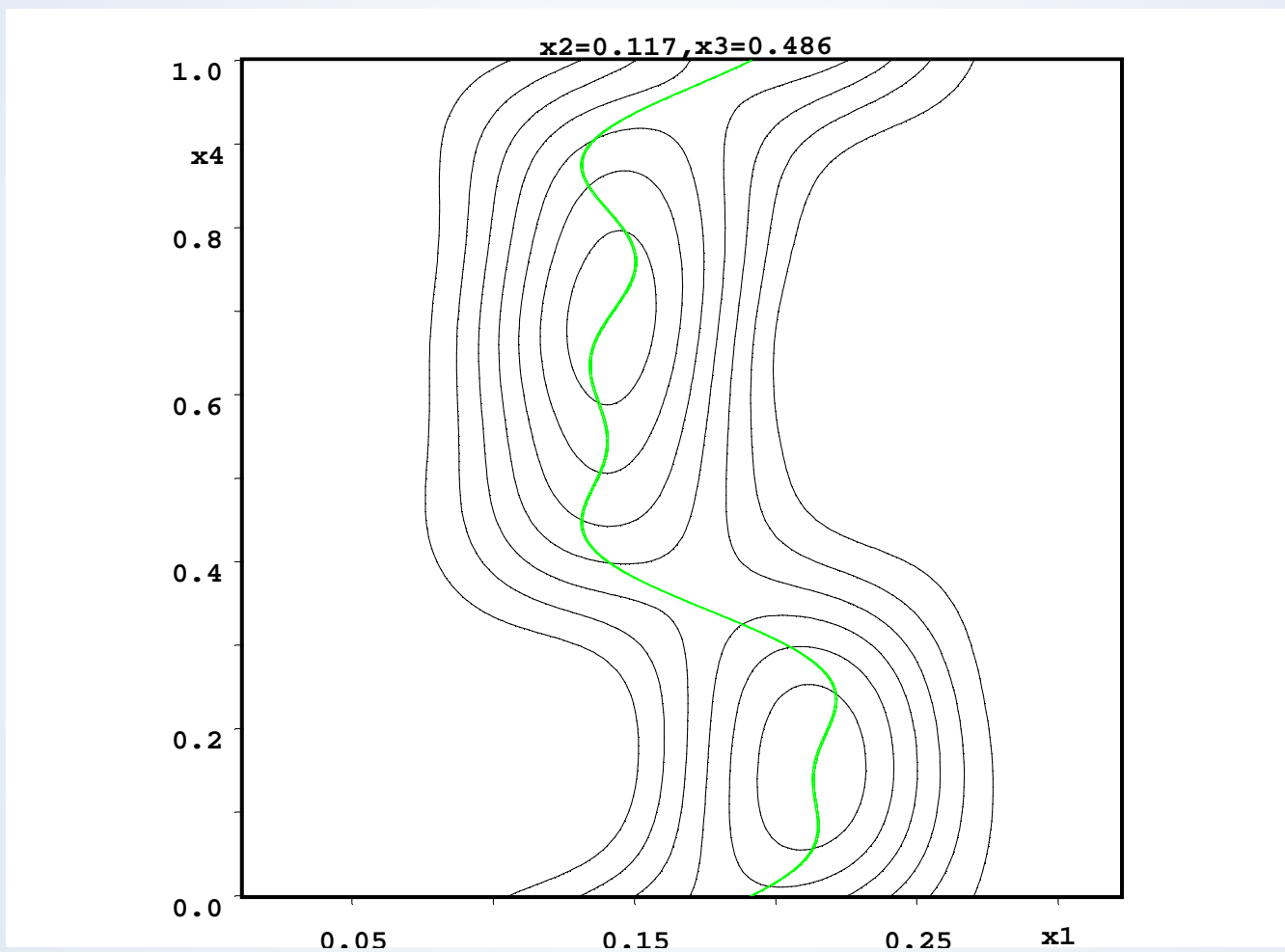
G. Oszlányi and A. Süto, (2004). *Acta Cryst.* **A60**, 134.

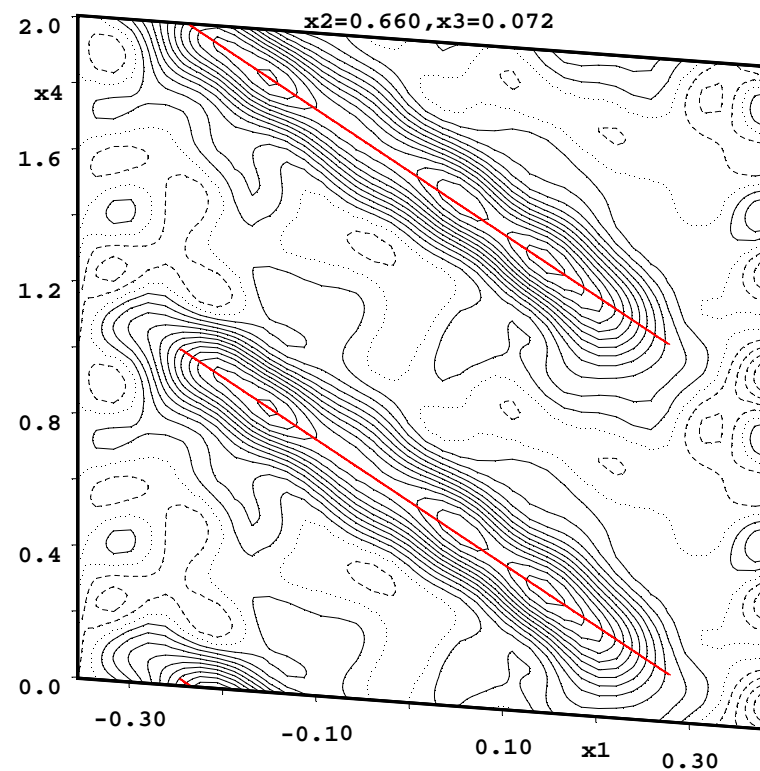
### For modulated structures:

L. Palatinus (2004). *Acta Cryst.* **A60**, 604.

L. Palatinus and G. Chapuis, (2007). *J. Appl. Cryst.* **40**, 786.

It represents a small revolution in the structure analysis of modulated crystals. It is applicable even for strongly modulated crystals having discontinuously modulated occupancies or positions.

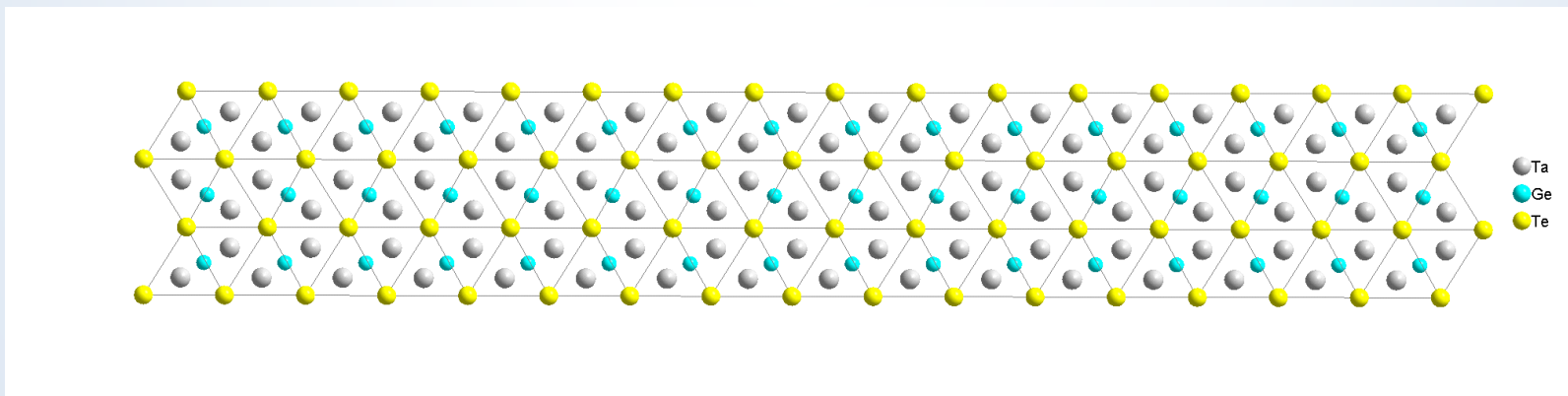




The modulation curves and corresponding density maps play a crucial role during the solution and refinement process. But a fine presentation should be made in the 3d real space. Several cells and more different sections are to be presented to see various configurations in the modulated crystal:

The same example as for the crenel function -  $\text{TaGe}_{0.354}\text{Te}$  - F. Boucher, M. Evain & V. Petříček, (1996). *Acta Cryst.*, B52, 100:

Average structure

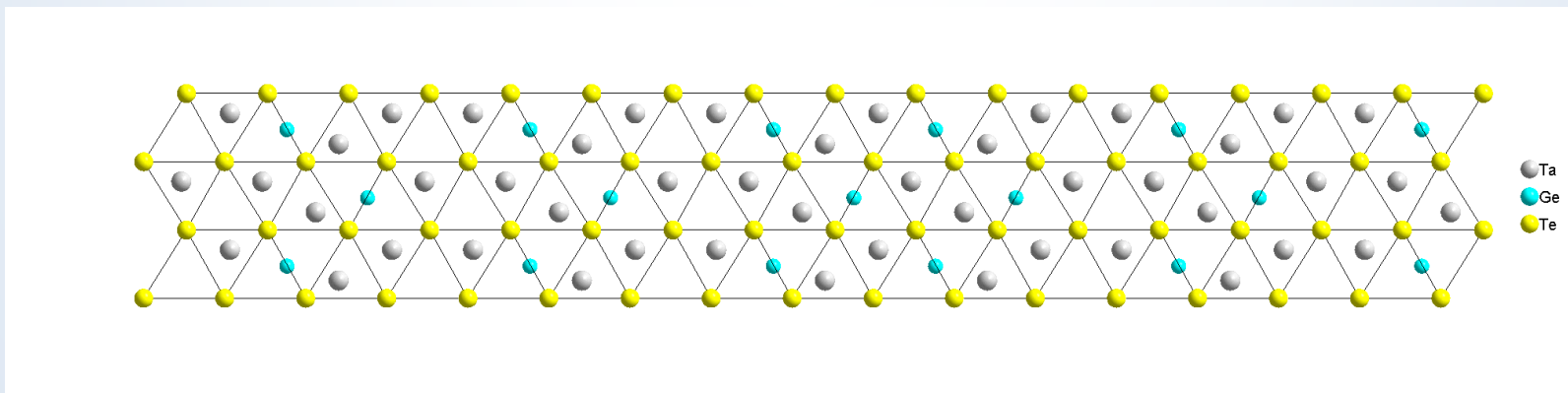




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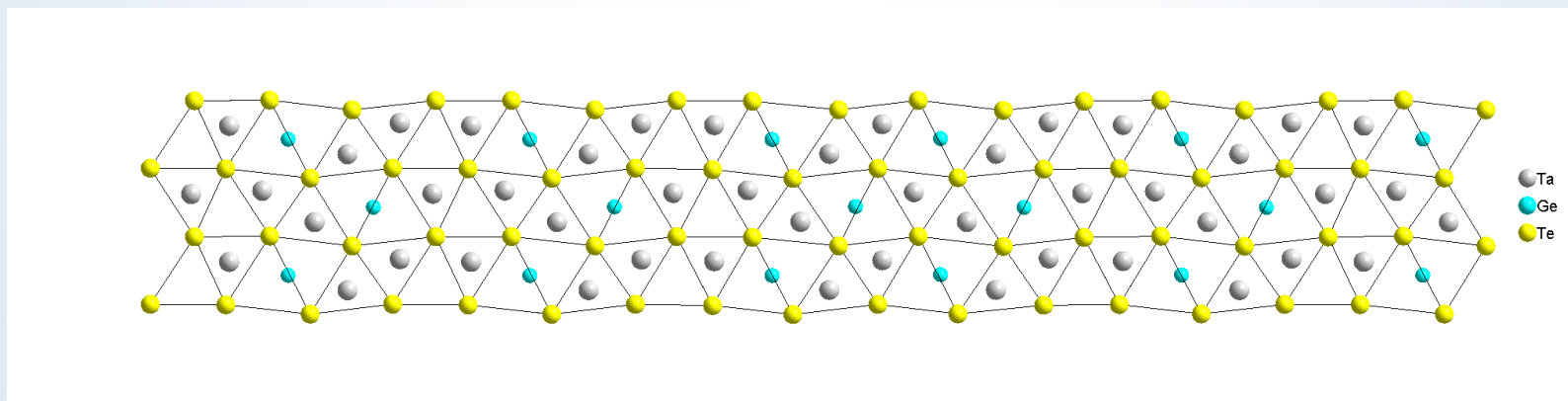
Only occupational modulation



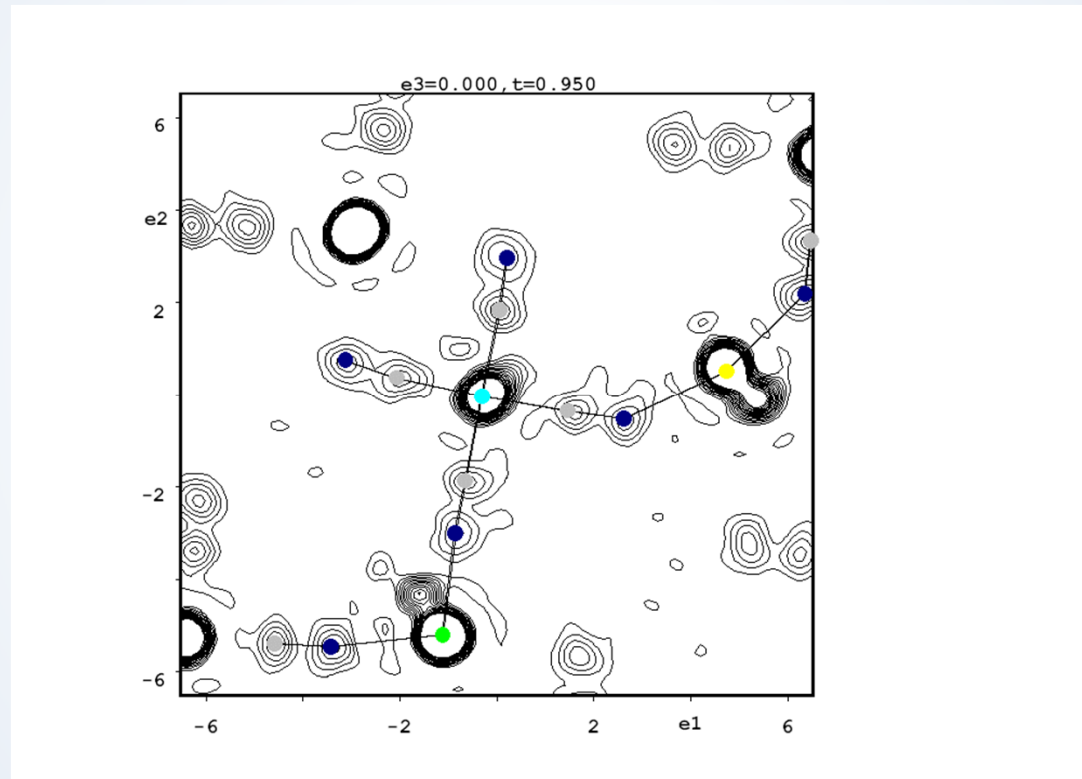
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The same example as for the crenel function -  $\text{TaGe}_{0.354}\text{Te}$  - F. Boucher, M. Evain & V. Petříček, (1996). *Acta Cryst.*, B52, 100:

Final result

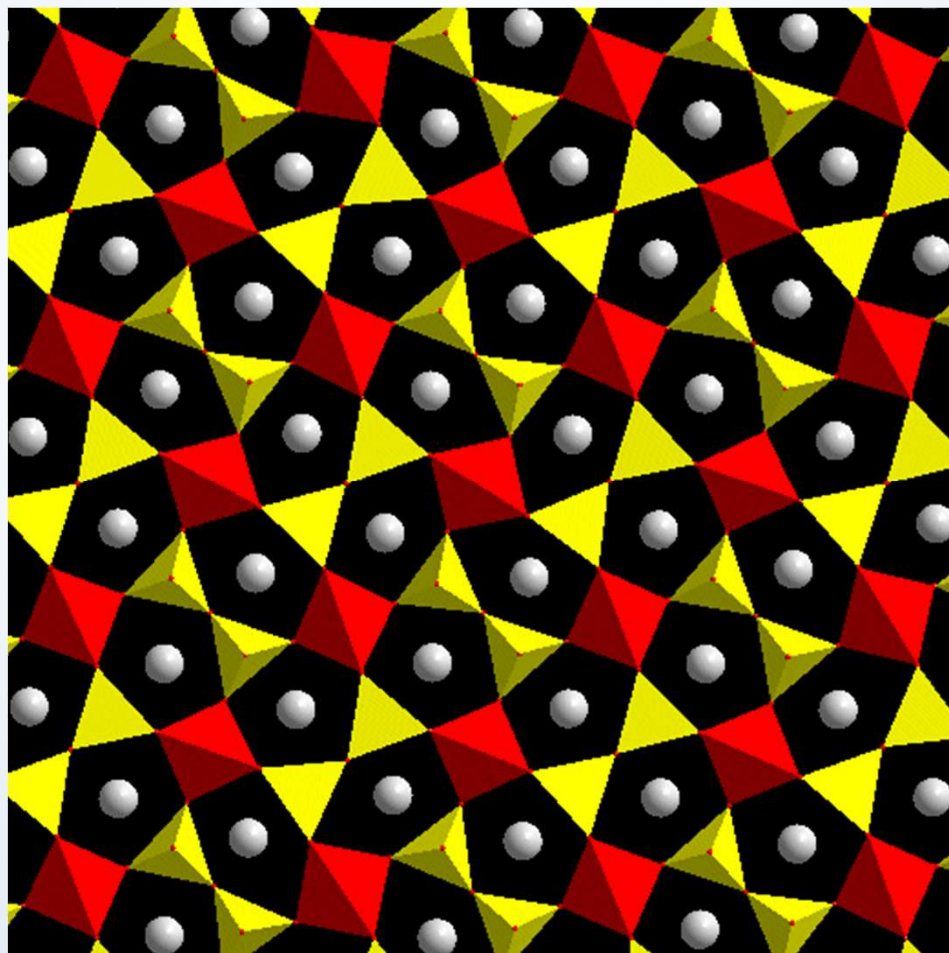


Another way how to present different situations in the crystal is an animation technique.



here combined with the density map.

L. Bindi, P. Bonazzi, M. Dušek, V. Petříček and G. Chapuis, *Acta Cryst.* (2001).  
B57, 739-746.



## Shubnikov space groups



- Magnetic moment – axial vector
- Any symmetry operation can be combined with “time inversion” -  $\theta$ , which inverts the magnetic moment

Symmetry operation:  $\hat{S} = (\mathbf{R}, \theta | \mathbf{s})$

$\mathbf{R}$  rotational part (proper or improper)  
 $\theta$  time inversion  
 $\mathbf{s}$  translational part

$$\mathbf{r}_j = \mathbf{R}\mathbf{r}_i + \mathbf{s} \quad \mathbf{M}_j = \theta \det(\mathbf{R}) \mathbf{R} \mathbf{M}_i$$

Set of symmetry operators of the magnetic crystal  $\rightarrow$  Shubnikov space group.

## Shubnikov space groups



Type I:  $F$  – Fedorov space groups

All symmetry operation have  $\theta=1 \rightarrow$  magnetic ordering allowed, magnetic reflections could be detected

e.g.  $Pmna$

Type II:  $F + F \cdot 1'$  gray space groups  $\rightarrow$  no magnetic ordering allowed, no magnetic reflections could be detected

e.g.  $Pmna1'$

Type III:  $D + (F - D) \cdot 1'$  where  $D$  is equi-translational subgroup of  $F$  of the order 2 half of symmetry elements have  $\theta=1$ , half  $\theta=-1$ . All translations have  $\theta=1$ .

e.g.  $Pm'n'a$

Type IV:  $D + (F - D) \cdot 1'$  where  $D$  is equi-class subgroup of  $F$  of the order 2

There translation vectors having  $\theta=-1$ . Reflections violating the original translation symmetry will appear as pure magnetic reflections – ordering effect is directly detectable.

e.g.  $P_a m'n'a$  (BNS) alternatively  $P_{2c} m'mn$  (OG)



## Shubnikov space groups



**Representation analysis:**

Representation	Dimension	Shubnikov space group	Axes	Origin shift	
Tau1	1	P[c]-6m2	(1,0,0 0,1,0 0,0,2)	(0,0,0)	Details
Tau2	1	P[c]-6c2	(1,0,0 0,1,0 0,0,2)	(0,0,0)	Details
Tau3	1	P[c]-6c2	(1,0,0 0,1,0 0,0,2)	(0,0,1/4)	Details
Tau4	1	P[c]-6m2	(1,0,0 0,1,0 0,0,2)	(0,0,1/4)	Details
Tau5	2	P[b]m	(0,1,0 0,0,2 1,0,0)	(0,1/4,0)	Details
Tau6	2	P[b]m	(0,1,0 0,0,2 1,0,0)	(0,0,0)	Details

## Shubnikov space groups



**List of kernels and epikernels:**

Shubnikov space group	Axes	Origin shift	Representation
P[c]-6m2	(1,0,0 0,1,0 0,0,2)	(0,0,0)	Tau1
P[c]-6c2	(1,0,0 0,1,0 0,0,2)	(0,0,0)	Tau2
P[c]-6c2	(1,0,0 0,1,0 0,0,2)	(0,0,1/4)	Tau3
P[c]-6m2	(1,0,0 0,1,0 0,0,2)	(0,0,1/4)	Tau4
A[a]ma2	(0,0,2 1,1,0 -1,1,0)	(0,0,0)	Tau5
A[a]mm2	(0,0,2 1,1,0 -1,1,0)	(1/4,0,0)	Tau5
A[a]mm2	(0,0,2 1,1,0 -1,1,0)	(0,0,0)	Tau6
A[a]ma2	(0,0,2 1,1,0 -1,1,0)	(1/4,0,0)	Tau6
P[b]m	(0,1,0 0,0,2 1,0,0)	(0,1/4,0)	Tau5
P[b]m	(0,1,0 0,0,2 1,0,0)	(0,0,0)	Tau6

Select from above kernels/epikernels one representative of a family of Shubnikov space groups for testing.



## Shubnikov superspace groups

The superspace approach developed by de Wolff, P.M., Janssen T. & Janner, A. (1981). *Acta Cryst.*, **A37**, 625-636 can be used to describe periodical perturbations, commensurate or incommensurate with the basic 3 dimensional translation symmetry.

For magnetic structures :

Petříček V., Fuksa J. and Dušek M. *Acta Cryst.* (2010). **A66**, 649-655.

Perez-Mato J.M., Ribeiro J.L., Petříček V. and Aroyo, M.I. *J.Phys.: Condens. Matter* (2012). **24**, 163201.

$$\mathbf{M}_i = \mathbf{M}_{i,0} + \sum_m \mathbf{M}_{i,ms} \sin 2\pi m \mathbf{k} \mathbf{r}_0 + \sum_m \mathbf{M}_{i,mc} \cos 2\pi m \mathbf{k} \mathbf{r}_0$$

## Shubnikov superspace groups

Superspace symmetry operation:

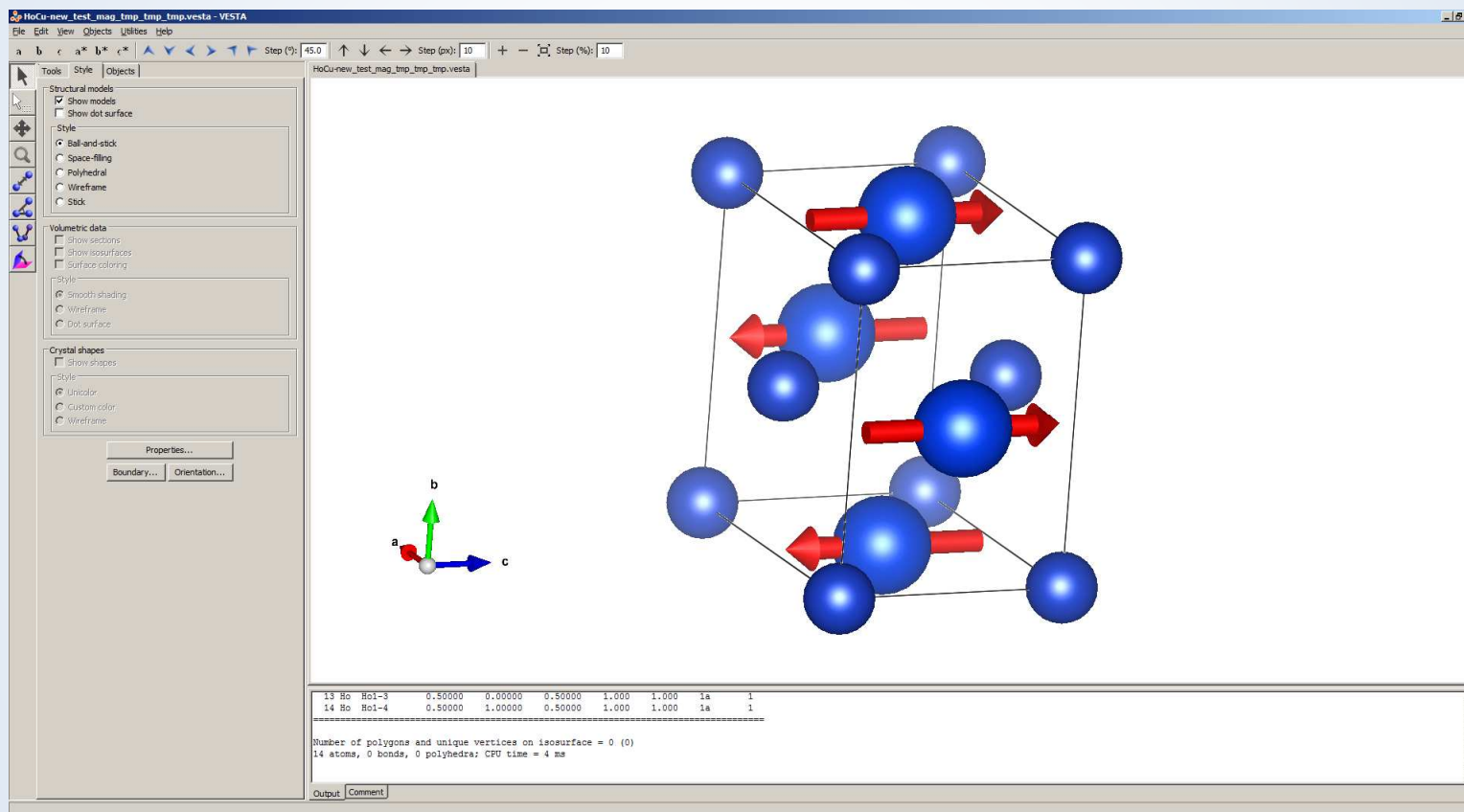
$$\hat{\mathcal{S}} = \left( \begin{bmatrix} \mathbf{R}_E & 0 \\ \mathbf{R}_M & \mathbf{R}_I \end{bmatrix}, \theta | \mathbf{s} \right)$$

Where  $\mathbf{R}_E$ ,  $\mathbf{R}_M$  and  $\mathbf{R}_I$  are external  $3 \times 3$ , mixed  $1 \times 3$ , internal  $1 \times 1$  block matrices and the time inversion sign, respectively. The right-upper  $3 \times 1$  block is a zero column.

non-zero matrix blocks are related:  $\mathbf{R}_M = \mathbf{k}\mathbf{R}_E - \mathbf{R}_I\mathbf{k}$

Time inversion of two types:  $1': 0 = (\mathbf{E}, -1 | 0, 0, 0, 0)$  and  $1': s = (\mathbf{E}, -1 | 0, 0, 0, 1/2)$

## Visualization of magnetic structure with VESTA program (Koichi Momma and Fujio Izumi)



## ISODISTORT – H.T. Stokes, B. J. Campbell and D. M. Hatch

**ISODISTORT: distortion**

Space Group: 194 P6<sub>3</sub>/mmc D6h-4, Lattice parameters: a=5.63760, b=5.63760, c=24.15760, alpha=90.00000, beta=90.00000, gamma=120.00000  
 Default space-group preferences: monoclinic axes a(b)c, monoclinic cell choice 1, orthorhombic axes abc, origin choice 2, hexagonal axes, SSG standard setting  
 Ba1 4f (1/3,2/3,z), z=-0.04041, Ba2 4f (1/3,2/3,z), z=0.36991, Ba3 2b (0,0,1/4), Co4 2a (0,0,0), Co5 4e (0,0,z), z=0.39525, Co6 4f (1/3,2/3,z), z=-0.17818, Cl1 2c (1/3,2/3,1/4), O2 12k (x,2x,z), x=0.16063, z=-0.14926, O3 12k (x,2x,z), x=-0.14584, z=-0.04948, O4 2d (1/3,2/3,3/4)

Include magnetic Co distortions  
 k point: GM, k16 (0,0,0)  
 IR: mGM2+, mk16i3  
 P1 (a) 194.270 P6<sub>3</sub>/mm<sup>3</sup>c', basis={(0,-1,0),(1,1,0),(0,0,1)}, origin=(0,0,0), s=1, i=2, k-active= (0,0,0)  
 Lattice parameters of undistorted supercell: a=5.63760, b=5.63760, c=24.15760, alpha=90.00000, beta=90.00000, gamma=120.00000

View distortion  View diffraction  CIF file  Distortion file  Domains  Primary order parameters  Modes details  Complete modes details

TOPAS.STR  IR matrices

Enter mode and strain amplitudes: 

P6<sub>3</sub>/mmc[0,0,0]mGM2+ (a) 194.270 P6<sub>3</sub>/mm<sup>3</sup>c', basis={(0,-1,0),(1,1,0),(0,0,1)}, origin=(0,0,0), s=1 i=2, ferromagnetic, k-active= (0,0,0)

[Co4.a]A2g(a)  
 [Co5.e]A2(a)  
 [Co6.f]A2(a)

P6<sub>3</sub>/mmc[0,0,0]GM1+ (a) 194.264 P6<sub>3</sub>/mmc s=1 i=1

Parameters: 

"View distortion":  
 Atomic radius:  Angstroms  
 Maximum bond length:  Angstroms  
 Length of magnetic moment vectors:  Angstroms/magneton

Applet width:  pixels

Supercell enlargement:  x  x

"View distortion" and "View diffraction":  
 Maximum displacement per mode:  Angstroms  
 Maximum strain per mode:   
 Maximum occupancy per mode:   
 Maximum magnetic moment per mode:  magnetons  
 Include strain modes in TOPAS.STR:  yes  no

Use alternate (possibly nonstandard) setting in CIF file  
 Basis vectors of subgroup lattice relative to parent:  
 a' =  a +  b +  c  
 b' =  a +  b +  c  
 c' =  a +  b +  c  
 Origin of subgroup relative to parent:  a +  b +  c

Password for beta version:

## Bilbao Crystallographic Server – M. Aroyo, J.M. Pérez-Mato

### Sections

[Retrieval Tools](#)

[Magnetic Space Groups](#)

[Group-Subgroup](#)

[Representations](#)

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[Material from the ITOOnline Workshop  
\(September 2011\)](#)



[Material from the school on the  
server \(June 2009\)](#)

<b>MAXSUB</b>	Maximal Subgroups of Space Groups
<b>SERIES</b>	Series of Maximal Isomorphic Subgroups of Space Groups
<b>WYCKSETS</b>	Equivalent Sets of Wyckoff Positions
<b>NORMALIZER</b>	Normalizers of Space Groups
<b>KVEC</b>	The k-vector types and Brillouin zones of Space Groups
<b>SYMMETRY OPERATIONS</b>	Geometric interpretation of matrix column representations of symmetry operations

### Magnetic Space Groups

<b>MGENPOS</b>	General Positions of Magnetic Space Groups
<b>MWYCKPOS</b>	Wyckoff Positions of Magnetic Space Groups
<b>MAGNEXT</b>	Extinction Rules of Magnetic Space Groups
<b>MAXMAGN</b>	Maximal magnetic space groups for a given space group and a propagation vector

### Group - Subgroup Relations of Space Groups

<b>SUBGROUPGRAPH</b>	Lattice of Maximal Subgroups
<b>HERMANN</b>	Distribution of subgroups in conjugated classes
<b>COSETS</b>	Coset decomposition for a group-subgroup pair
<b>WYCKSPLIT</b>	The splitting of the Wyckoff Positions
<b>MINSUP</b>	Minimal Supergroups of Space Groups
<b>SUPERGROUPS</b>	Supergroups of Space Groups
<b>CELLSUB</b>	List of subgroups for a given k-index.
<b>CELLSUPER</b>	List of supergroups for a given k-index.
<b>NONCHAR</b>	Non Characteristic orbits.
<b>COMMONSUBS</b>	Common Subgroups of Space Groups
<b>COMMONSUPER</b>	Common Supergroups of Two Space Groups





**Jana2006** is a crystallographic program focused to solution, refinement and interpretation of difficult, especially modulated structures. It calculates structures having up to three modulation vectors from powder as well as single crystal data measured with X-ray or neutron diffraction. The input diffraction data can be unlimitedly combined, the combination of powder neutron data with single crystal X-ray data being a typical example. The structure solution can be done using the built-in charge flipping algorithm or by calling an external direct methods program. Jana can handle multiphase structures (for both powder and single crystal data), merohedric twins as well as twins with partial overlap of diffraction spots, commensurate and composite structures. It contains powerful transformation tools for symmetry (group-subgroup relations), cell parameters and commensurate-super-cell relations. Wide scale of constraints and restraints is available including a powerful rigid body approach and possibility to define a local symmetry affecting only part of the structure. The latest development of Jana concerns magnetic structures.

[More about Jana2006](#) (PowerPoint presentation, 10MB)

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Academy of Sciences | Institute of Physics  
Dept of Structure Analysis | Laboratory of Crystallography  
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CRYSTALLOGRAPHIC COMPUTING SYSTEM FOR STANDARD AND MODULATED STRUCTURES

*Vaclav Petricek, Michal Dusek & Lukas Palatinus*

## News

**June 12, 2012 Workshop in Klatovy, Czech Republic.** Half-day workshop at the colloquium of the Czech and Slovak Crystallographic Association about solution of 3d and 4d structures in Jana2006, in Czech language. Contact: [dusek@fzu.cz](mailto:dusek@fzu.cz).

**February 23 - March 08, 2012 Workshops in Kosice, Slovakia.** Four half-day workshops about solution of 3d structures in Jana2006, in Czech language. Dates: 23.2., 1.3., 6.3., 8.3. Contact: [dusek@fzu.cz](mailto:dusek@fzu.cz).

**July 01, 2011 Maintenance release of Jana2006 for Windows** can be obtained in [Download area](#). It cumulates small improvements before larger changes planned for the summer 2011.

**May 03, 2011 Maintenance release of Jana2006 for Windows** can be obtained in [Download area](#). It cumulates various improvements and bug fixes. [More ...](#)

**October 23, 2010 Maintenance release of Jana2006 for Windows** can be obtained in [Download area](#). It cumulates various improvements and bug fixes. [More ...](#)

**April 1, 2010 Maintenance release of Jana2006 for Windows** can be obtained in [Download area](#). It cumulates various improvements and bug fixes. [More ...](#)

**January 26, 2010 The Third Ad Hoc Workshop on Jana2006**, 8-9 March 2010. [More ...](#)

**January 06, 2010 The Second Ad Hoc Workshop on Jana2006**, 18-19 Jan 2010. [More ...](#)

**January 05, 2010 International school on aperiodic crystals.** [More ...](#)

**October 21, 2009 Maintenance release of Jana2006 for Windows** can be obtained in [Download area](#). It cumulates various improvements and bug fixes. [More ...](#)

**September 16, 2009 Ad hoc workshops on Jana.** We offer informal two days workshops on topics selected by participants. [More ...](#)

**October 23, 2008 Maintenance release of Jana2006 for Windows** can be obtained in [Download area](#). It cumulates various improvements and bug fixes. [More ...](#)

**October 14, 2008 Eset NOD32 antivirus slows down Jana2006.** [More ...](#)

**[Ad hoc workshops on Jana \[Next: 11, 12\]](#)**

## Ad hoc workshops on Jana

Ad hoc workshops are small two days workshops organized in Praha by authors of Jana software. A new workshop is usually organized when we find an overlap in topics in the list of preliminary registrations for at least five people. In such case the program details and the dates are discussed with participants and fixed. Another participants can register for a workshop with fixed program and date up to 12-15 people. Participants pay themselves all their expenses and they are expected to bring their own laptops with Windows. However, we can lend one or two laptops to those who cannot bring their own ones. The registration fee is zero (in Czech Crowns). Workshop certificates are issued by our skilled artists on demand.

### Topics

- (INT1) Introduction to Jana software, solution of simple 3d structures
- (INT2) Introduction to modulated structures, solution of simple 4d structures
- (PWD) Refinement of powder data
- (PROF) Powder profile functions
- (TW) Twins and multiphases of 3d and modulated structures
- (CHF) Application of charge flipping to 3d and modulated structures
- (DIS) Disorder, mixed sites, application of rigid body refinement for disordered structures
- (RIG) Rigid body refinement and application of local symmetry
- (GEOM) How to keep geometry
- (LOC) Application of local symmetry
- (CRENEL) Discontinuous modulation functions (crenel and sawtooth)
- (COMM) Commensurate structures
- (COMP) Composite structures (i.e. samples with two coexisting lattices)
- (FIVE) Five dimensional structures
- (MAG) Magnetic structures
- (ELD) Electron diffraction
- any other topic (please specify)

**Registration** By e-mail to Michal Dusek (dusek@fzu.cz). For workshops without fixed program or for future not yet announced workshops please give a list of selected topics, rated by importance. Although a workshop can usually have only four topics (one topic takes 1/2 of day) the number of selected topics can be higher, it helps finding overlaps. Please indicate if you have already experience with Jana software. New users should select the topic "INT1". Those who would like to participate in an advanced workshop (without INT1) but don't have an experience with Jana should make Example 1.1 and 2.1 of the [cookbook](#) as a homework.

### Passed workshop:

- 1st Jana2006 ad hoc workshop
- 2nd Jana2006 ad hoc workshop
- 3rd Jana2006 ad hoc workshop
- 4th Jana2006 ad hoc workshop
- 5th Jana2006 ad hoc workshop (Advanced)
- 6th Jana2006 ad hoc workshop
- 7th Jana2006 ad hoc workshop (MAG)
- 8th Jana2006 ad hoc workshop
- 9th Jana2006 ad hoc workshop
- 10th Jana2006 ad hoc workshop
- 11th Jana2006 ad hoc workshop
- 12th Jana2006 ad hoc workshop
- 13th Jana2006 ad hoc workshop (PWD)
- 14th Jana2006 ad hoc workshop (MAG)
- 15th Jana2006 ad hoc workshop (ELD)
- 16th Jana2006 ad hoc workshop
- 17th Jana2006 ad hoc workshop (Basic)
- 18th Jana2006 ad hoc workshop (Adv)
- 19th Jana2006 ad hoc workshop (ELD)
- 20th Jana2006 ad hoc workshop (MAG)
- 21st Jana2006 ad hoc workshop (Basic)
- 22nd Jana2006 ad hoc workshop (Advanced)

**Forthcoming workshops:** See [here](#)

**REQUIREMENTS for workshops** This link serves for collecting requirements for future Jana2006 Ad Hoc workshops. As soon as an overlap in topics is found a new Ad Hoc Workshop is organized.

For more details: <http://jana.fzu.cz/>  
Praha – Prague (Czech Republic)

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**The program is freeware:** result of scientific research made with a close collaboration with Jana users

Work is continuously supported by Academy of Sciences of the Czech Republic and (occasionally) by Grant agency of the Czech Republic

*End of the first part*

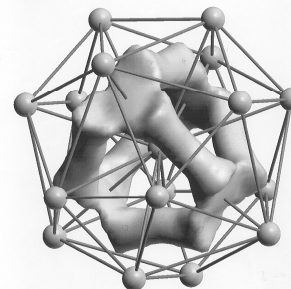
# Jana2006 – powder option

Václav Petříček, Jan Rohlíček and Michal Dušek  
Institute of Physics  
Academy of Sciences of the Czech Republic  
Praha

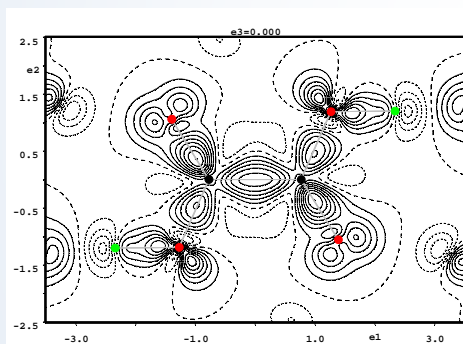


Jana2006 is a refinement program originally developed for single crystal data. Except for the refinement of basic structural parameters it allows also:

ADP parameters till 6<sup>th</sup> order – **ionic conductors, positional disorder**

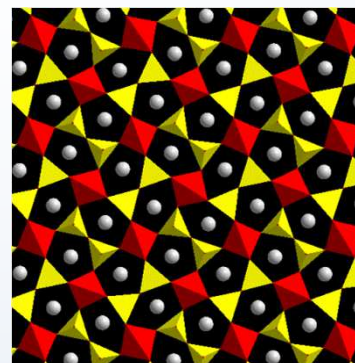


E.Gaudin, F.Boucher,  
V.Petříček, F.Taulelle and  
M.Evain, Acta Cryst. (2000),  
**B56, 402-408.**



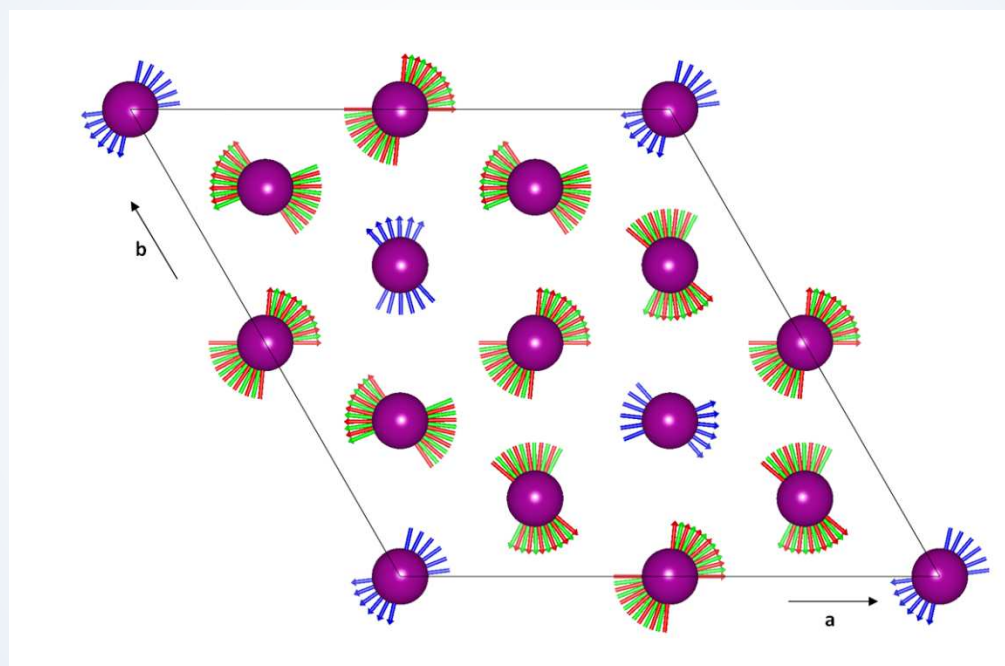
**Charge density studies - multipole refinement**

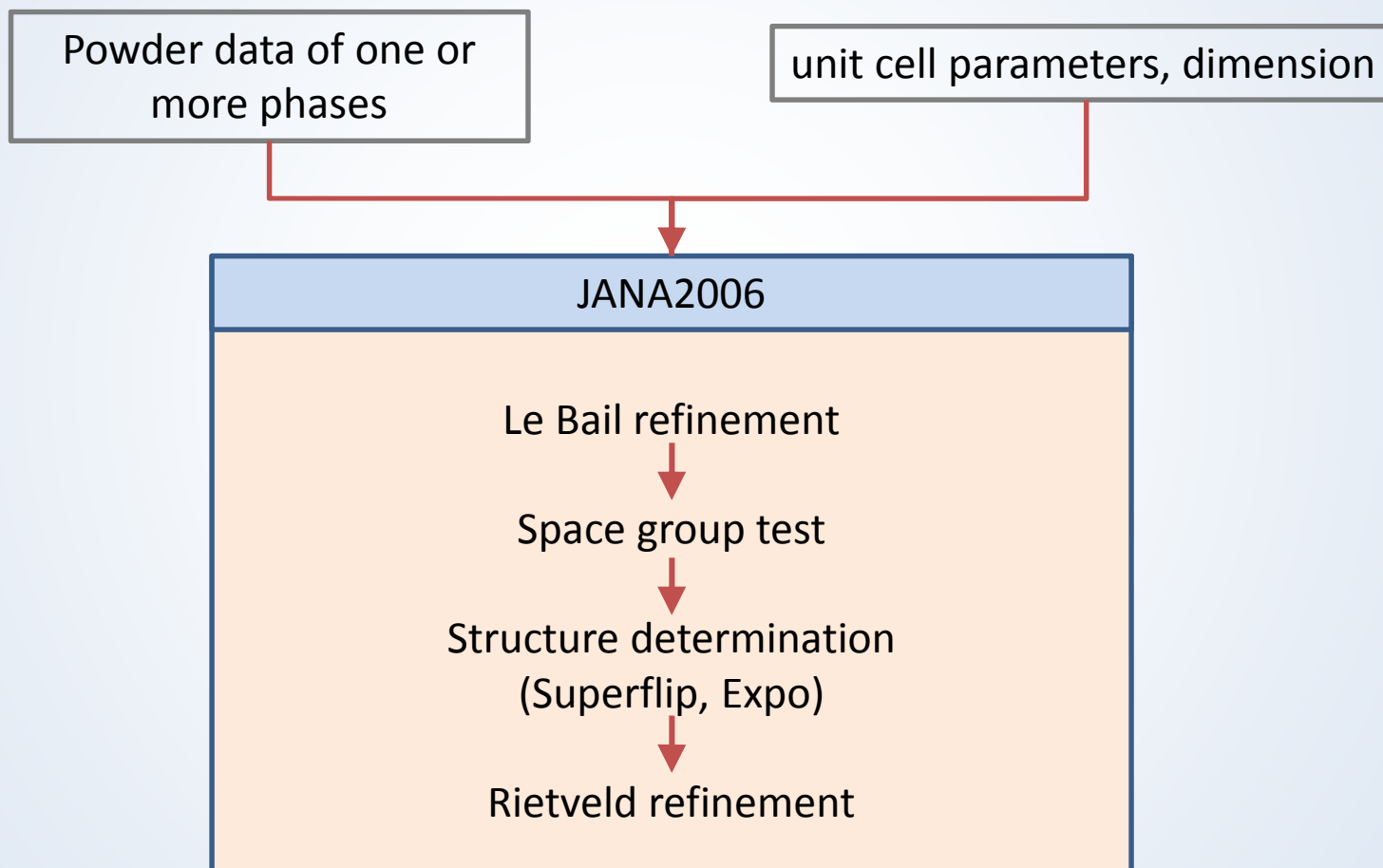
Refinement of **modulated** and **composite** structures



Melilite – Bindi L., Bonazzi P.,  
Dušek M., V.Petříček and  
Chapuis G., (2001), Acta  
Cryst., **B57, 739-746.**

## magnetic structures





## Le Bail refinement

Separate intensities from powder pattern  
in **Jana2006** activated by  
„Make only profile matching“



## Structure solution

Use separated intensities to solve the  
crystal structure



## Rietveld refinement

Refine the model from the structure  
solution process by using powder data

## „Edit Profile“:

**Profile functions:** Gauss, Lorentz and pseudo-Voigt

**Background corrections:** Legendre, Chebyshev polynomials, “cos” functions

**Peak asymmetry:** Simpson, Berar-Baldinozzi, Finger-Cox-Jephcoat  
fundamental approach (Cheary-Coelho)

**Preferred orientation:** March-Dollase, Sasa-Uda

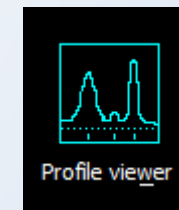
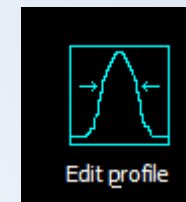
**Roughness:** Pitchke-Hermann-Matter, Suortti

**Anisotropic line-broadening** according to Stephens modified by Leineweber and Petricek to include modulated phases

## „Profile Viewer“:

Edit manual background

checking calculated and measured data



The powder diffraction option (Rietveld refinement + leBail technique) was implemented in **2001**: Dušek, M., Petříček, V., Wunschel, M., Dinnebier, R., E. & Van Smaalen, S. (2001). J. Appl. Cryst. 34, 398-404.



**Gaussian:**  $G(b_G, x) = \frac{1}{\sqrt{2\pi b_G}} \exp(-x^2 / 2b_G^2)$

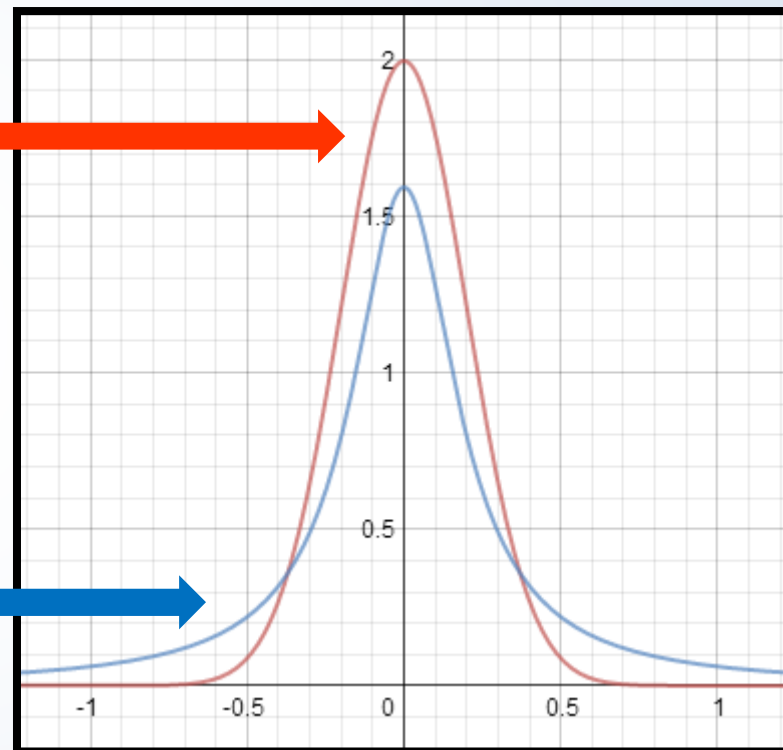
$$b_G^2 = \frac{H_G^2}{8 \ln 2}$$

where  $H_G$  is Full-Width at Half-Maximum

**Lorentzian:**  $L(b_L, x) = \frac{2}{\pi b_L} \frac{1}{1 + (2x/b_L)^2}$

$$b_L = H_L$$

where  $H_L$  is Full-Width at Half-Maximum



Voigt function: 
$$V(b_G, b_L, x) = \int_{-\infty}^{+\infty} G(b_G, x') L(b_L, x - x') dx'$$

is a convolution of Gaussian and Lorentzian function. For powder profile we use a simpler analytical approximation of the Voigt function called pseudo-Voigt function:

$$pV(H, x) = \eta L(H, x) + (1 - \eta) G(H, x)$$

the parameters  $\eta$  and  $H$  are functions of  $H_G$  and  $H_L$

$$H^5 = H_G^5 + 2.69269 H_G^4 H_L + 2.42843 H_G^3 H_L^2 + 4.47163 H_G^2 H_L^3 + 0.07842 H_G H_L^4 + H_L^5$$

$$\eta = 1.36603 \frac{H_L}{H} - 0.47719 \left( \frac{H_L}{H} \right)^2 + 0.11116 \left( \frac{H_L}{H} \right)^3$$

The diffraction line broadening induced by the sample is usually divided into to crystalline size effect  $T$  and microstrain  $Y$ . In a first approximation they have the following angular dependence:

$$H_L = b_L = X_L / \cos \theta + Y_L \tan \theta \quad X_L = 180\lambda / \pi T$$

Similar equations are valid for Gaussian distribution but then an additional factor  $8 \ln 2$  is to be applied to get  $b_G$  from  $H_G$ .

$$H_G^2 = 8 \ln 2 b_G^2 = X_G^2 / \cos^2 \theta + Y_G^2 \tan^2 \theta$$

These ideal equations are valid only if broadening, induced by experimental conditions are neglected. This means we can use the refined values to make conclusions about crystalline size and microstrain only if some type of fundamental approach or correction for experimental broadening is used.

But usually we are using an additional terms to take into account other experimental effects and then direct interpretation of refined parameters is not possible.

For the Gaussian term the formula Cagliotti, Pauletti & Ricci, 1958 (Nucl.Instrum., **3**, 223) is used:

$$b_G^2 = U \tan^2 \theta + V \tan \theta + W + \frac{P}{\cos^2 \theta}$$

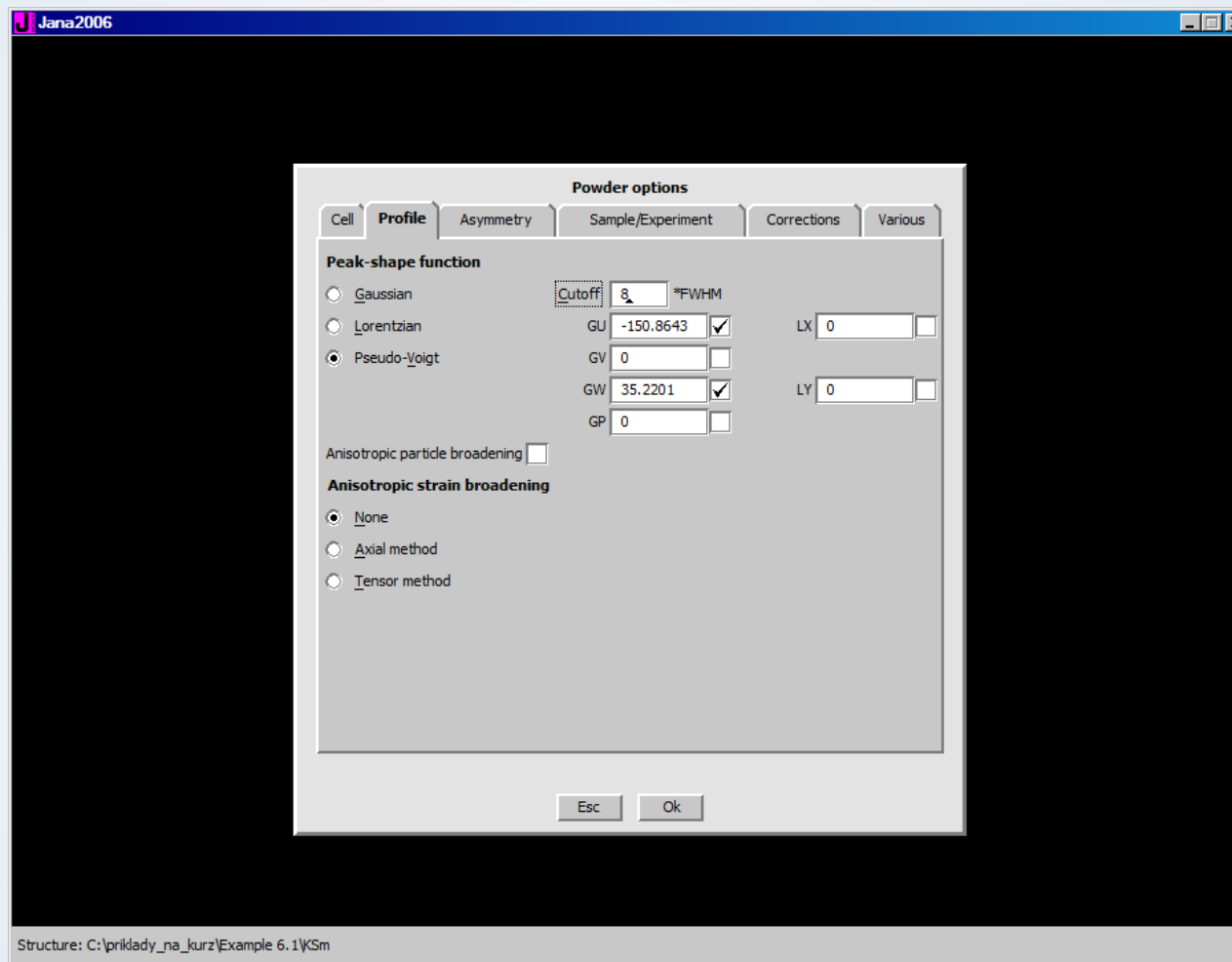
In the original formula only three first terms were used. The last one was introduced later as a Scherrer term and it is connected with “crystalline size”. But from the fact that:

$$\tan^2 \theta + 1 = \frac{1}{\cos^2 \theta}$$

follows that only two of three coefficients  $U, W, P$  can be refined simultaneously.

For the Lorentzian part we use the same terms as for pure crystal broadening but, as mentioned above, they cannot be used directly to find sample characteristics.

$$b_L = X_L / \cos \theta + Y_L \tan \theta$$



The angular parameters (LX, LY,...) are measured in 0.01 deg, the squared ones in 0.0001 deg<sup>2</sup>.

This phenomenological model is based on a general tensor expression in which the anisotropic strain is described by a symmetrical 4th order tensor:

$$\sigma^2(hkl) = D^{ijmn} h_i h_j h_m h_n = \sum_{H,K,L} S_{HKL} h^H k^K l^L$$

The first term is just a general tensor expression where Landau summation convention is used and which allows a simple derivation of symmetry restrictions similar to those for 4th order ADP parameters. The second term has an explicit form as used by Stephens in which summation is restricted to  $H + K + L = 4$

This term is used to modify equations for  $b_G$  and  $b_L$

For modulated structures this method has been generalized by A. Leineweber and V. Petříček, (2007). J. Appl. Cryst. ,**40**, 1027-1034.

Simpson's method – Peak is combined with several shifted peaks having the identical shape according to the formula:

$$P_{corr}[2\theta] = \frac{1}{6n} \sum_{i=1}^{2n+1} k_i P \left[ 2\theta + a \left( \frac{i-1}{2n} \right)^2 \cot 2\theta \right]$$

$$k_i = 1 \quad \text{for} \quad i = 1 \text{ or } i = 2n + 1$$

$$k_i = 4 \quad \text{for} \quad i = 2m$$

$$k_i = 2 \quad \text{for} \quad i = 2m + 1 \quad m \neq 0 \quad m \neq n$$

$a$  is the only parameter to be refined

## Berar-Baldinozzi method

$$P_{corr}(2\theta) = P(2\theta) \left( 1 + \frac{p_1 F_1(z) + p_2 F_2(z)}{\tan \theta} + \frac{p_3 F_1(z) + p_4 F_2(z)}{\tan 2\theta} \right)$$

where  $F_1(z) = H_1(2z) \exp(-z^2)$        $z = \frac{\theta - \theta_0}{FWHM}$

$$F_2(z) = H_3(2z) \exp(-z^2)$$

$H_n(z)$  stand for Hermit polynomials



By axial divergence – according to Finger, Cox and Jephcoat, (1994) J.Appl.Cryst. **27**, 892-900.

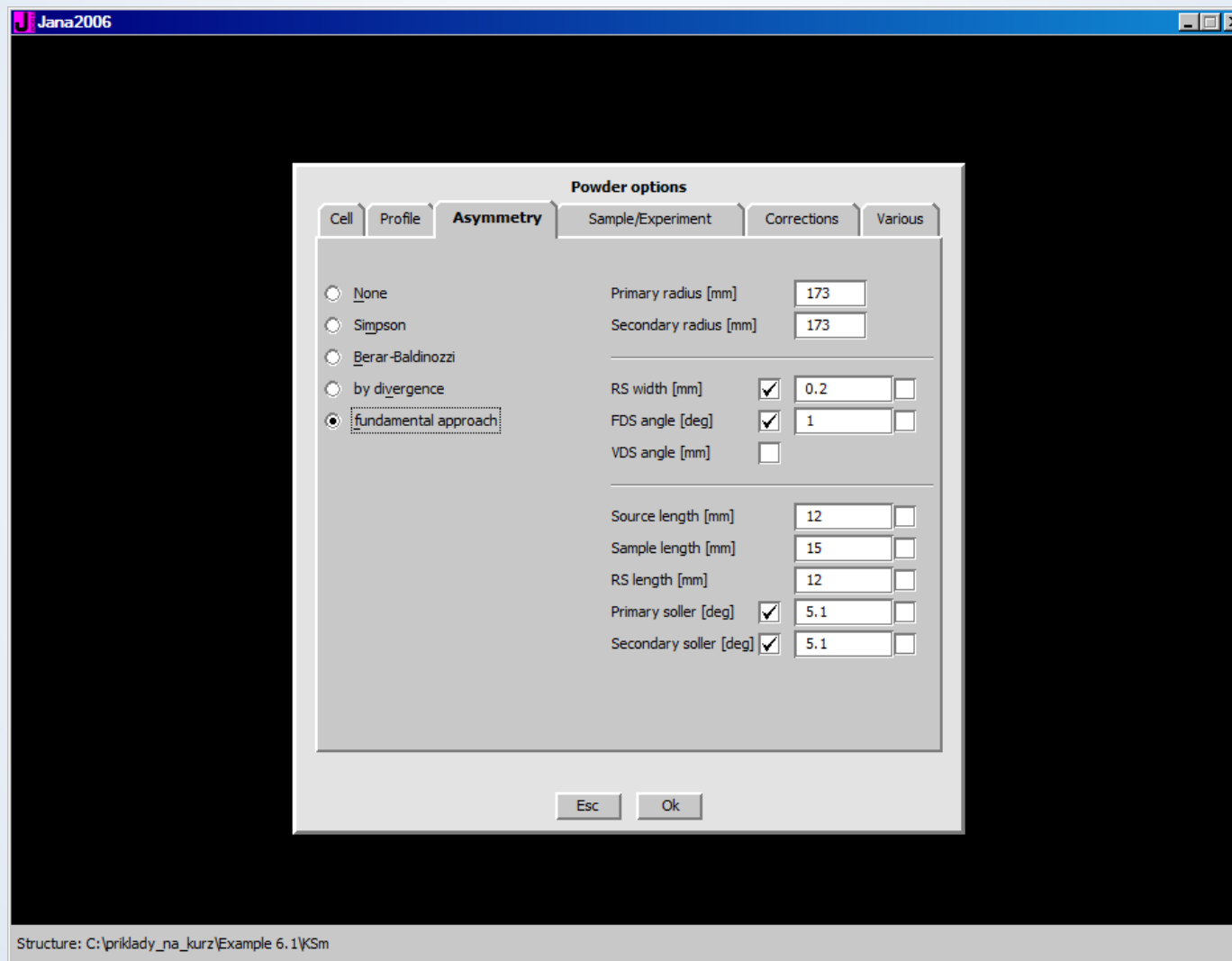
$$H/L \quad S/L$$

These parameters are strongly correlated. Our recommendation is to estimate their ratio and keep it as a restriction during the refinement.

Fundamental approach – it follows the method introduced by Cheary and Coelho, (1998), J.Appl.Cryst. **31**, 851-861.

This method can estimate the profile asymmetry just on the base of experimental parameters. For Bragg-Brentano geometry it works very nicely.

The method makes multiple convolution of several functions.



Legendre polynomials – set of orthonormal polynomials defined on the interval  $\langle -1, +1 \rangle$ . The measured interval have to be first linearly projected into this interval. Then their orthonormality considerably suppress their mutual correlations.

Chebyshev polynomials – they are also orthogonal but in difference sense

$$\int_{-1}^1 w(x) p_i(x) p_j(x) dx = \delta_{ij}$$

Legendre polynomials:  $w(x) = 1$

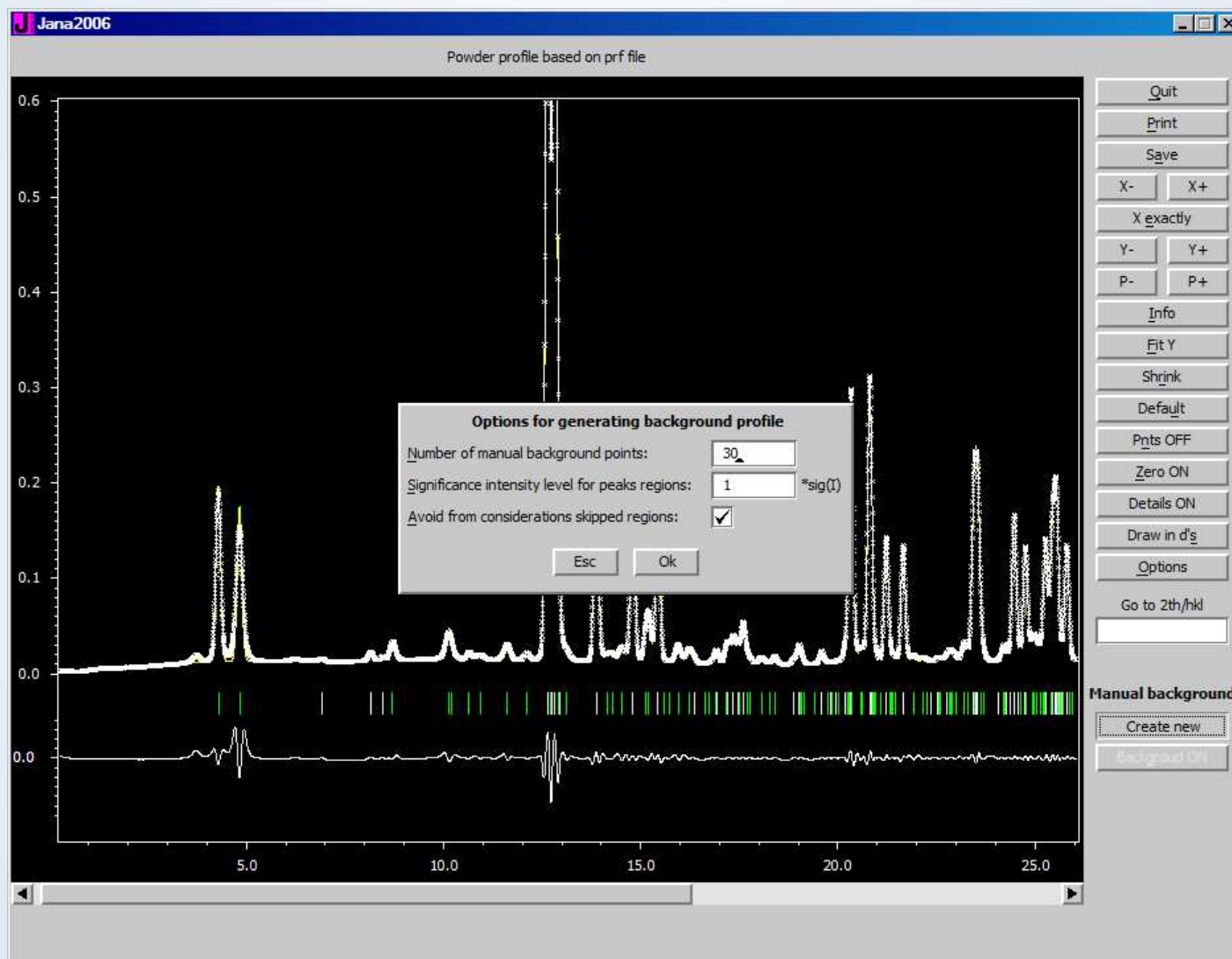
Chebyshev polynomials:  $w(x) = 1/\sqrt{1-x^2}$

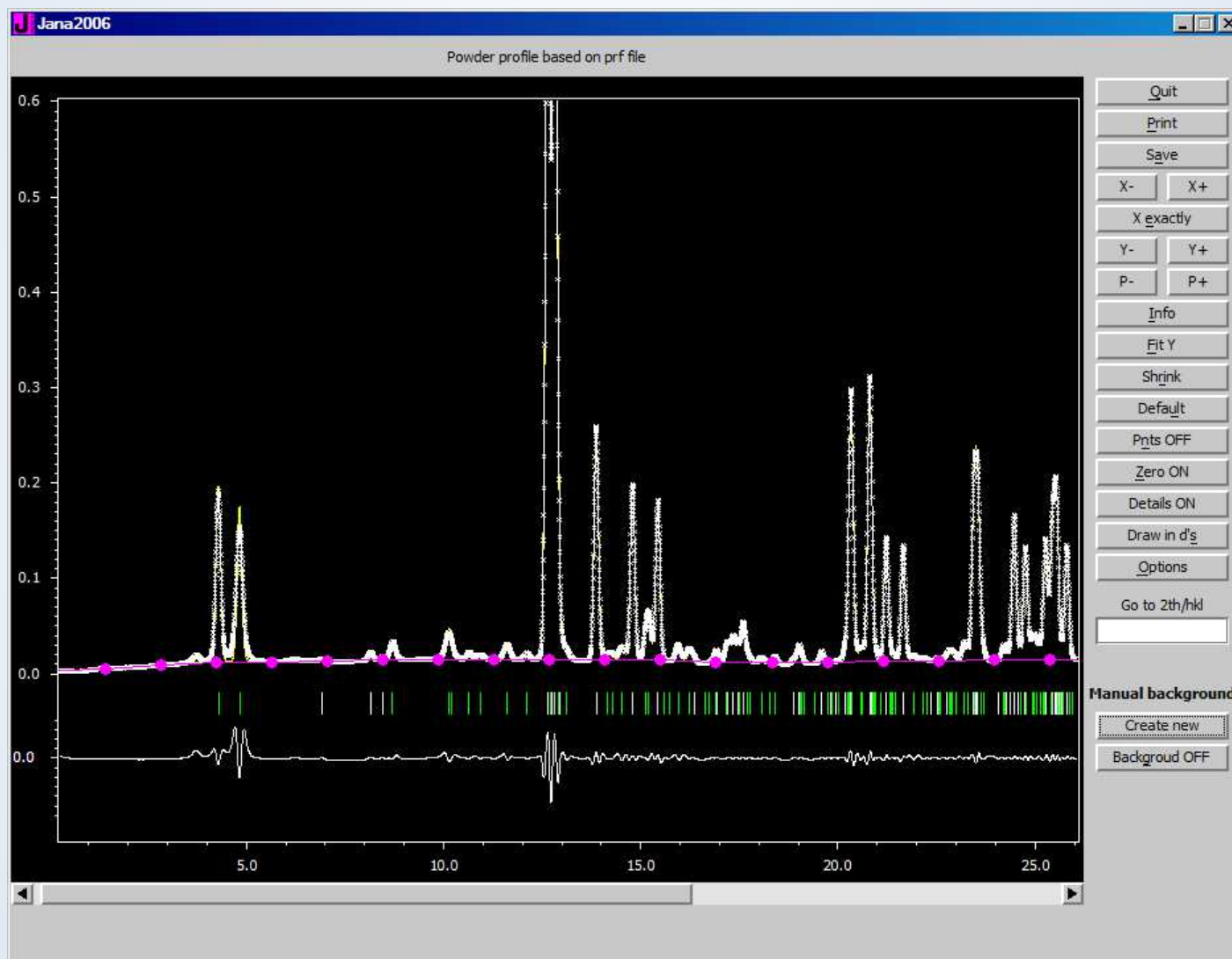
$$w(x) = \sqrt{1-x^2}$$

For data collected uniformly along diffraction interval we should preferably use the unique weight.

Manual background – the background is expressed as a set of background intensities over the diffraction interval. The actual value is calculated by a linear interpolation. This method can very effectively describe even very complicated background profiles. But it needs some user assistance to select it properly. Moreover, this first background estimation can be combined with some of the previous continuous functions.

In the program Jana2006 there is a tool to select a first estimation of manual background, but then you can interactively modify it.







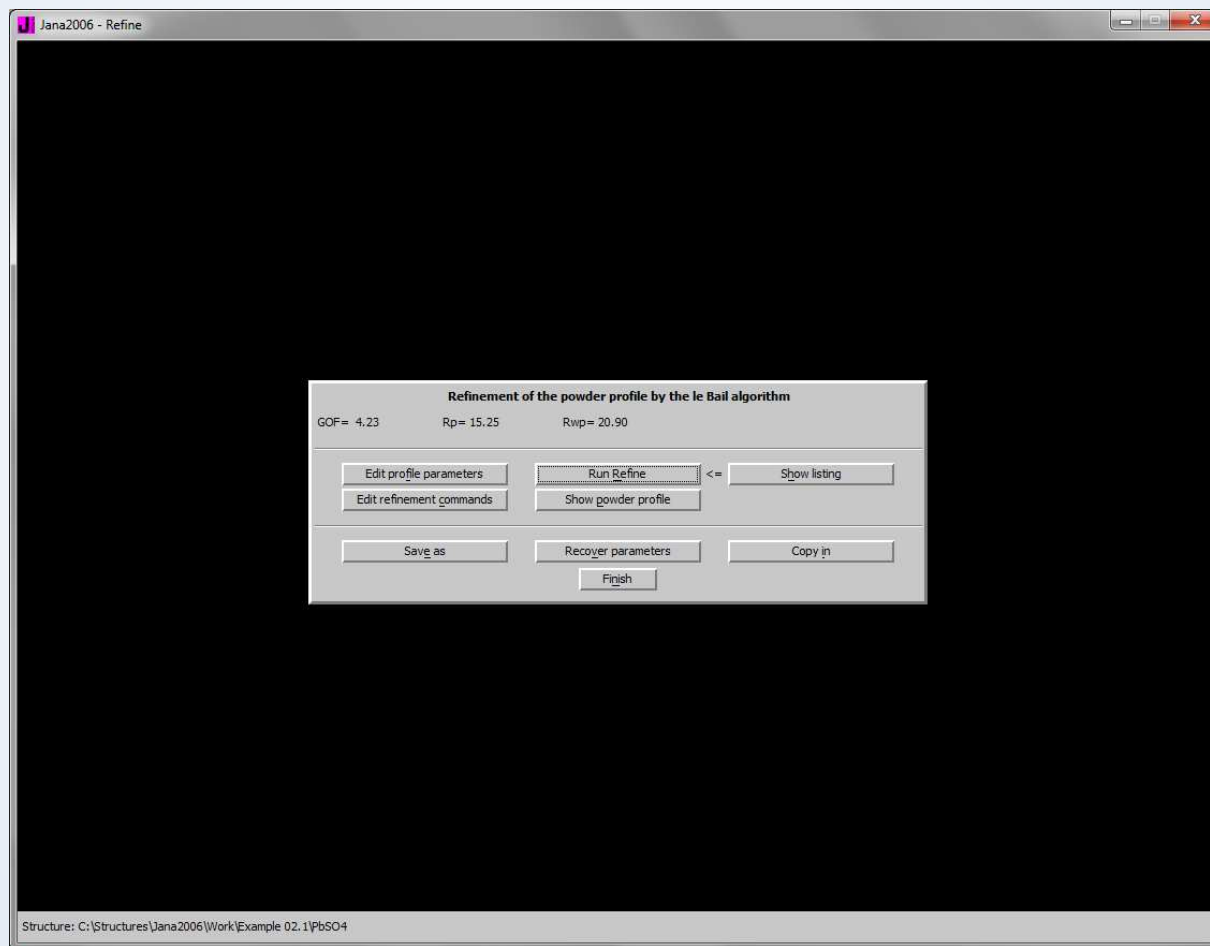
shift – it defines the zero shift (again in units of 0.01 deg). This value is to be added to the theoretical peak position to get a position in experimental profile

sycos – in analogy with the Fullprof:  $\Delta 2\theta = c \cdot \cos \theta$   
is connected with a specimen displacement

ysin – in analogy with the Fullprof:  $\Delta 2\theta = s \cdot \sin 2\theta$   
is connected with a transparency correction

These three corrections are combined together to the actual peak position.

Based on a peak decomposition and subsequent refinement of profile parameters.



In Jana2006 is also used to predict symmetry by comparing of profile fits for different space groups:

**Select cell centering**

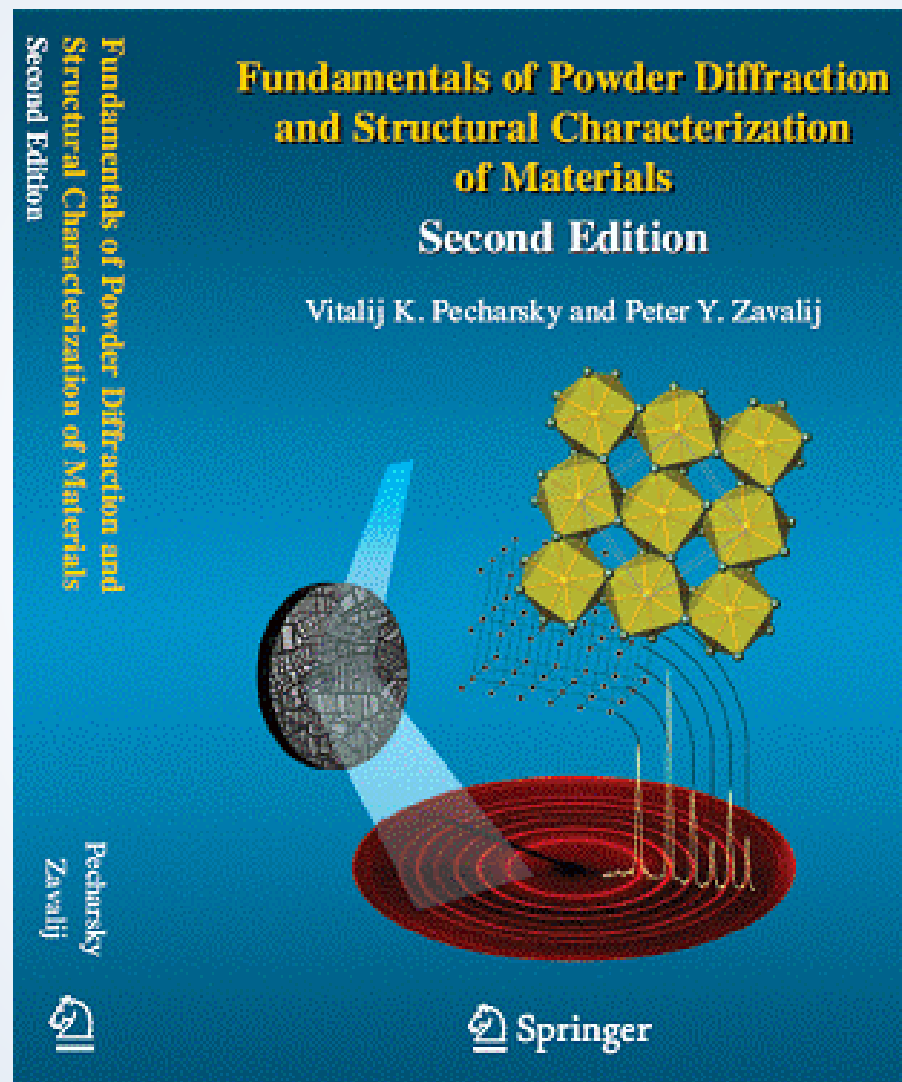
	Centering	Rp(obs)/Rp(all)	N(Extinct)/N(Gener)
<input checked="" type="radio"/>	P	5.767/6.001	0.0000
<input type="radio"/>	A	29.780/28.383	0.4957
<input type="radio"/>	B	32.304/30.561	0.5043
<input type="radio"/>	C	32.435/30.742	0.4964
<input type="radio"/>	I	32.483/30.731	0.4978
n.a.	R-obverse	15.042/14.608	0.6655
n.a.	R-reverse	15.042/14.608	0.6655
<input type="radio"/>	F	45.853/43.206	0.7482

Warning: The cell centering need not be one you expect from collection as the program first transform the cell to the reduced form.  
Moreover after your selection the program makes another transformation whenever the centering is not the standard one.

In Jana2006 is also used to predict symmetry by comparing of profile fits for different space groups:

Space group	Rp(obs)/Rp(all)	N(Extinct)/N(Gener)	FOM
Pn21a	5.621/5.874	0.1332	5.4828
Pnma	5.621/5.874	0.1332	5.4828
Pnmm	5.584/5.838	0.0632	5.6540
Pnm21	5.584/5.838	0.0632	5.6540
Pn21m	5.584/5.838	0.0632	5.6540
Pmma	5.797/6.028	0.0700	5.8170
Pm2a	5.797/6.028	0.0700	5.8170
P21ma	5.797/6.028	0.0700	5.8170
P212121	5.790/6.024	0.0271	5.9421
P22121	5.766/6.002	0.0158	5.9542
P21221	5.787/6.020	0.0203	5.9591
P21212	5.782/6.014	0.0181	5.9595
P2221	5.763/5.998	0.0090	5.9711
P2212	5.758/5.992	0.0068	5.9714
P2122	5.779/6.010	0.0113	5.9765
P222	5.755/5.988	0.0000	5.9883

Buttons: Back, Next, Cancel



*Thank you for your attention*

Example 2.1 – PbSO<sub>4</sub> ... simple regular structure, solution, refinement

Example 2.2 – Y<sub>2</sub>O<sub>3</sub> ... different asymmetry corrections

Example 6.1 – KSm ... simple modulated structure